

# Two-dimensional semiconductors for transistors

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**Abstract** | In the quest for higher performance, the dimensions of field-effect transistors (FETs) continue to decrease. However, the reduction in size of FETs comprising 3D semiconductors is limited by the rate at which heat, generated from static power, is dissipated. The increase in static power and the leakage of current between the source and drain electrodes that causes this increase, are referred to as short-channel effects. In FETs with channels made from 2D semiconductors, leakage current is almost eliminated because all electrons are confined in atomically thin channels and, hence, are uniformly influenced by the gate voltage. In this Review, we provide a mathematical framework to evaluate the performance of FETs and describe the challenges for improving the performances of short-channel FETs in relation to the properties of 2D materials, including graphene, transition metal dichalcogenides, phosphorene and silicene. We also describe tunnelling FETs that possess extremely low-power switching behaviour and explain how they can be realized using heterostructures of 2D semiconductors.

Field-effect transistors (FETs) — three-terminal systems consisting of source, drain and gate electrodes — are integral in many electronic devices, allowing them to achieve energy-efficient high-speed switching. Semiconducting materials, also known as the channel of a FET, span the source and drain electrodes. The channel is electrically isolated from the gate electrode by the gate dielectric. The effective operation of FETs relies on efficient electrostatic coupling between the electric field induced by the gate voltage and the channel, without allowing electrons to flow between them.

Traditional FETs are, most often, based on bulk or 3D semiconductor channels composed of silicon and the III–V semiconductors, GaAs and GaN. These 3D materials have been successfully scaled down to nanoscale dimensions over the past 5 decades following Moore's law, which in 1965 predicted that the density of transistors in a chip will double every 2 years<sup>1–4</sup>. The semiconductor industry has established long-term strategies and research goals based on Moore's Law, but this prediction has recently started to falter and substitutes are under consideration<sup>5</sup>.

## Short-channel effects

The dimensions of FETs, and hence the lengths of the channels, continue to decrease in the quest for higher performance. Devices with shorter channel lengths have begun to experience high off-state currents; that is, some charge carriers are able to flow between the source and drain electrodes even on application of a gate voltage

that suppresses the flow of current. The presence of an off-state current increases the static power (the product of the drain voltage and off-state current). As a result, the reduction in size of FETs has been limited by the rate at which heat, caused by static power, is dissipated. The current leakage and the resulting challenges associated with heat dissipation caused by an increase in static power are collectively termed short-channel effects<sup>6,7</sup>. The consequences of short-channel effects are detrimental for device operation and energy efficiency to the extent that the International Technology Roadmap for Semiconductors (ITRS) predicts that transistor densities will double every 3 years rather than every 2 years<sup>8</sup>. To address these short-channel effects, devices such as multiple gate transistors<sup>9,10</sup>, FinFETs<sup>11,12</sup> and ultrathin body transistors (UTB)<sup>13–15</sup> are being actively pursued.

There is a need to identify new semiconductor materials that can mitigate short-channel effects. In addition, these materials should be compatible with existing complementary metal oxide semiconductor (CMOS) infrastructure. A starting point for identifying semiconductors for transistor channels is the examination of the FET electrostatics described by Poisson's equation. This equation yields a characteristic channel 'scaling length' given by  $\lambda = \sqrt{t_s t_b (\epsilon_s / \epsilon_b)}$ , where  $t_s$  is the semiconductor thickness,  $t_b$  is the gate dielectric thickness, and  $\epsilon_s$  and  $\epsilon_b$  are the semiconductor and gate insulator dielectric constants, respectively<sup>6,7</sup>. In reality, UTB semiconductors made from decreasing the thickness of 3D (bulk) materials suffer from dangling bonds, leading to scattering

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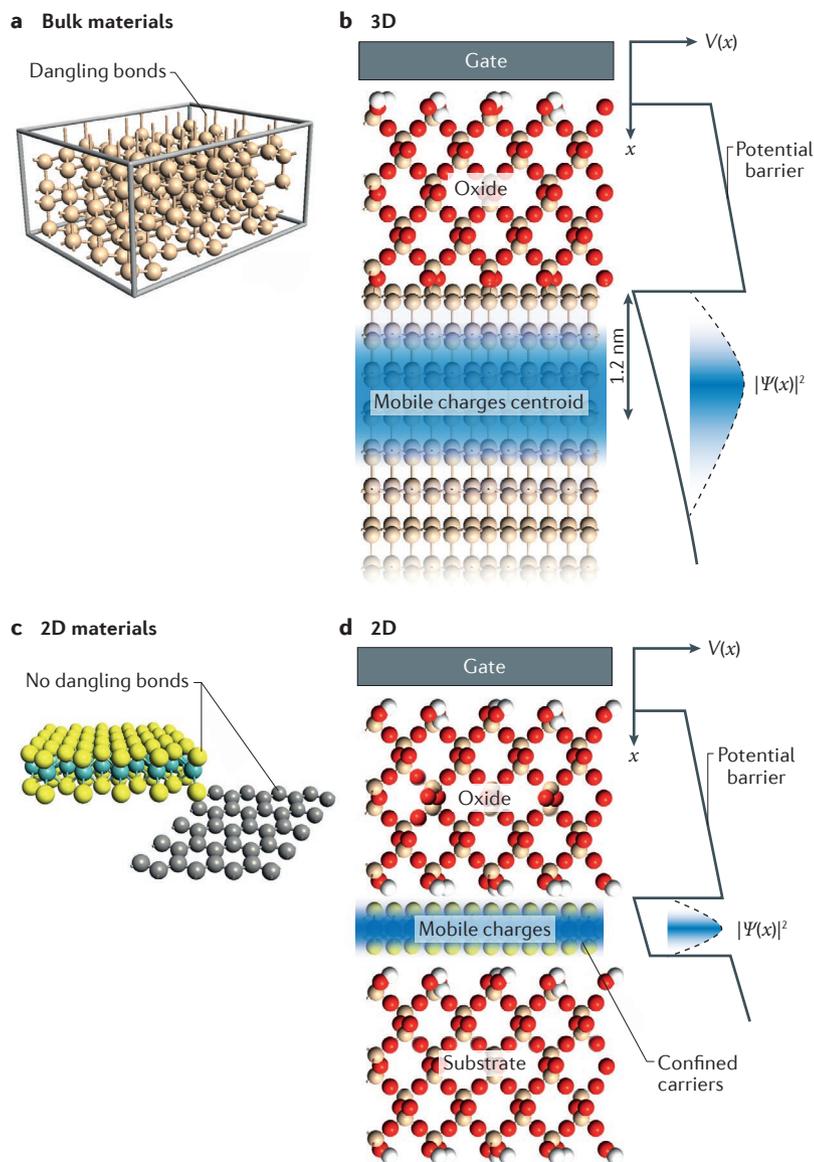
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of the charge carriers<sup>16,17</sup> (FIG. 1 a). As a result, the expected enhancement in gate electrostatic properties and substantial degradation of FET performance are observed. More specifically, the mobility,  $\mu$ , of charge carriers decreases with thickness to the sixth power,  $\mu \sim t^6$ , and the bandgap,  $E_g$ , increases by the square of the thickness,  $\Delta E_g \sim t^2$  (REF. 16). For transistors made from UTB 3D semiconductors, the substantial decrease in performance is observed because of the presence of dangling bonds, undesirable

coupling with phonons and the creation of interface states (that occur when the gate dielectric and the source/drain electrodes are deposited on top of the semiconductor). In 3D semiconductor short-channel FETs, current leakage arises from poor electrostatics between some electrons in the channel and the electric field applied by the gate (FIG. 1 b). However, in FETs that have a channel made from 2D semiconductors, all electrons are confined in naturally atomically thin channels and, hence, all carriers are uniformly influenced by the gate voltage<sup>17</sup> (FIG. 1 d). This excellent gate coupling allows the suppression of current leakage if a gate voltage is applied.



**Figure 1 | Advantages of 2D materials compared with 3D materials for FETs.**

**a** | Ultrathin 3D (bulk) semiconductors have dangling bonds that form traps for electrons and reduce the performance of field-effect transistors (FETs). **b** | Gate electrostatics and mobile charge distribution in 3D semiconductors.  $V(x)$  is the gate voltage as a function of the distance  $x$ . The potential barrier is the energy level of the gate dielectric and  $|\Psi(x)|^2$  refers to the probability function of the carriers in the semiconductors. The majority of the mobile charge carriers are located approximately 1.2 nm from the semiconductor gate dielectric interface. **c** | By contrast, 2D materials have pristine surfaces. **d** | In 2D materials, charge carriers are confined in the atomically thin semiconductor, resulting in a narrower mobile charge distribution. This confinement of charge carriers allows the carriers to be easily controlled by the gate voltage, leading to excellent gate electrostatics. Figure is adapted with permission from REF. 17, SPIE Journals and Proceedings.

## 2D materials

The range of 2D materials for application as the semiconducting component of FETs includes graphene, hexagonal boron nitride (h-BN), transition metal dichalcogenides (TMDs), silicene and phosphorene. Substantial research has been devoted to graphene, the 'original' 2D material, which has revealed interesting electronic<sup>18–21</sup> and photonic<sup>22,23</sup> phenomena. However, the absence of an energy bandgap makes graphene less desirable for use in FET switching settings that require high on-state currents but low off-state currents. Hence, the field has expanded into other 2D materials, predominantly semiconducting TMDs because they possess bandgaps in the range of 1–2 eV. The weak, van der Waals interactions between the layers of the corresponding bulk materials enable atomically thin layers of graphite, molybdenum disulfide ( $\text{MoS}_2$ ) and black phosphorus to be isolated relatively easily by exfoliation. This approach usually results in a low density of dangling bonds and minimal surface roughness (FIG. 1 c). The subsequent transfer of these 2D materials onto thin-layered insulators — for example, in the case of graphene transferred onto h-BN — almost completely eliminates dangling bonds in the insulator<sup>24</sup> and mitigates the effect of charged impurities by changing the dielectric environment in 2D semiconductors<sup>25</sup>. In another example, devices consisting of six-layered  $\text{MoS}_2$  (semiconductor) encapsulated by few-layered h-BN (gate insulator) show high mobilities, which reach  $34,000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  at low temperatures<sup>26</sup>. Recent progress suggests that such atomically thin 2D materials could be one pathway for electronic devices in the future<sup>27–29</sup>.

## Characteristic of FETs

To highlight the advantages of using 2D materials as semiconductors in FETs, the important characteristics of the devices must be covered first (FIG. 2). A typical FET is a three-terminal device consisting of a semiconducting channel between the source and drain electrodes. The current between the source and drain electrodes — passing through the semiconducting channel — is modulated by the application of a gate voltage. The transverse electric field created by the gate voltage can either deplete the channel of carriers so that no current flows between the source and drain electrodes (off-state), or enhance the concentration of carriers in the channel, allowing current to flow (on-state). Ideally, the off-state current is as small as possible and the on-state current is  $>10^4$  compared with the off-state current (that is, an on/off ratio of  $>10^4$ ).

A FET with a semiconductor channel composed of a single-layer TMD and its operating mechanisms are shown in FIG. 2. The gate barrier is an insulator of thickness  $t_b$ , the dielectric constant is  $\epsilon_b$  and the capacitance is  $C_b = \epsilon_b/t_b$ . The channel length along the  $x$  axis is  $L$  and the width along the  $y$  axis is  $W$ . Mobile electrons in the conduction (or valence) bands of the TMD layer form Bloch waves that must fit in the rectangular channel. This restricts the Bloch waves to  $k_x = n_x(2\pi/L)$  and  $k_y = n_y(2\pi/W)$ , where  $n_x, n_y = \dots, -1, 0, +1, \dots$  are integers, obtained by enforcing periodic boundary conditions, and  $k_x$  and  $k_y$  are wave vectors that represent the direction of electron wavefunction propagation in the crystal lattice. The density of mobile carriers in the 2D semiconductor sheet is given by summing all of the occupied allowed electron states over all of the spins and valleys:

$$n_s = (1/WL) \cdot \sum_{(k_x, k_y) \in \text{all spin, valley}} f(k_x, k_y) \quad (1)$$

The central problem of FET physics is determining how two voltages — one between the gate and source electrodes,  $V_{gs}$ , and the other between the drain and source electrodes,  $V_{ds}$  — control the occupation of the allowed channel modes,  $f(k_x, k_y)$ . The knowledge of this occupation function determines all of the output characteristics of the device. To solve this problem, we make three simplifications without neglecting important features of the FET. First, the gate insulator is ideal and allows no current to leak between the gate and the source and drain electrodes. Second, the contacts are ideal, which means there is no voltage drop at the source and drain contacts. Third, there is no scattering of electrons in the semiconductor channel. On the basis of these assumptions, we develop a detailed quantitative model for the operation of FET-based 2D materials (Supplementary information S1, S2 (box, figure)).

A FET based on 2D semiconductors has several key operating mechanisms<sup>30,31</sup>. When the source and drain electrodes are grounded ( $V_{ds} = 0$ ) and a positive voltage is applied on the gate electrode, the gate battery draws electrons from the gate metal and pumps the electrons into the semiconductor channel through the source and drain contacts. The physics of FET operation can be captured by understanding how the electron density of the mobile sheet,  $n_s$ , in the semiconductor channel depends on the gate voltage. The sheet charge is often assumed to be the gate capacitance,  $C_b = \epsilon_b/t_b$ , multiplied by the ‘excess’ voltage,  $qn_s = C_b(V_{gs} - V_T)$ , where  $q$  is charge and  $V_T$  is the threshold voltage determined by the difference in the work functions of the metal and the semiconductor. This is an approximation that is only true for a range of voltages when the transistor is deep in the on state — specifically, when  $V_{gs} - V_T \gg V_{th}$  (where the thermal voltage,  $V_{th} = kT/q$ , has a value of 26 mV at room temperature). This assumption fails near  $V_T$  and for off states of the FET, which may be appreciated by noticing that because  $n_s$  is the sheet density of filled states, it cannot be negative — however,  $V_{gs} - V_T < 0$  for  $V_{gs} < V_T$ .

The energy levels of the gate metal, insulator and 2D semiconductor along the  $z$  direction of the device are plotted in FIG. 2b. The relative position of the energy levels can

be represented by  $q\phi_B + qV_i - \Delta E_c + (E_{Fs} - E_c) = qV_{gs}$ , where  $q\phi_B$  is the metal–dielectric barrier height,  $qV_i$  is the voltage drop in the insulator,  $\Delta E_c$  is the conduction band offset,  $E_{Fs}$  is the quasi-Fermi level of the source electrode,  $E_c$  is the energy of semiconductor conduction band and  $qV_{gs}$  is the barrier modification by the applied gate voltage.

The carriers entering the channel through the source move to the right (red in FIG. 2c–f) and are in equilibrium with the energy level of the source electrode. The carriers entering the channel from the drain move to the left (blue in FIG. 2d–f) and are in equilibrium with the drain contact. The carriers cannot change direction in the channels because no scattering of electrons can occur. This results in a modal distribution of electrons in the 2D  $(k_x, k_y)$  space (FIG. 2c–f). By analysing the changes that occur in the energy levels in the presence of applied  $V_{gs}$  and  $V_{ds}$ , it is possible to derive the relationship between the carrier density and the gate voltage as,

$$e^{n_s/n_b} (e^{n_s/n_q} - 1) = e^{\frac{V_{gs} - V_T}{V_{th}}} \quad (2)$$

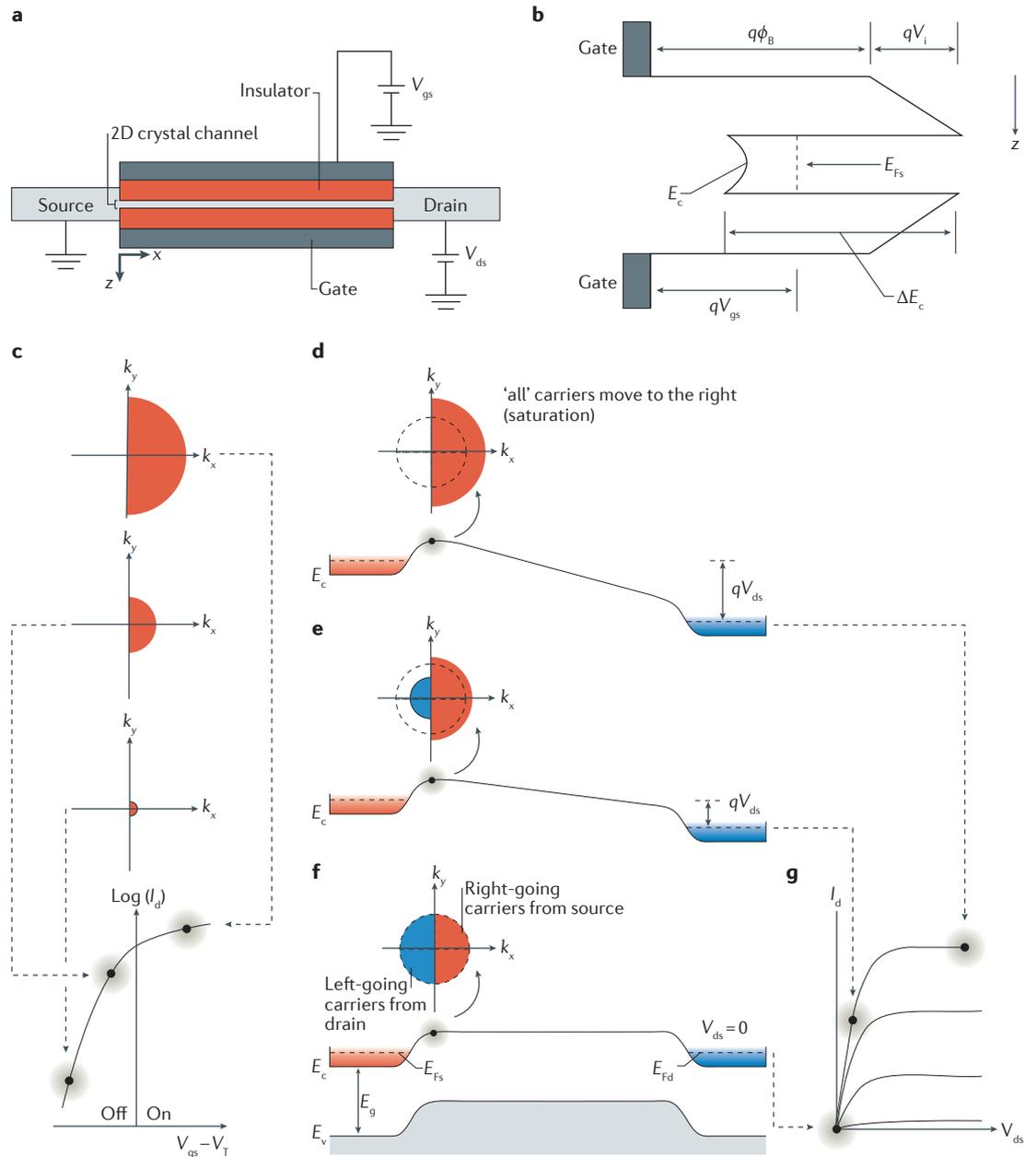
where  $n_s$  is the net carrier density,  $n_q$  is a characteristic 2D quantum concentration defined in the Supplementary information and  $n_b$  is the characteristic carrier density from the electrostatic capacitance (Supplementary information S1 (box)). This is a transcendental equation, the solution to which gives the mobile carrier density in the semiconductor channel as a function of the gate voltage,  $n_s(V_{gs})$ .

### Ideal FET output characteristics

We can use the solutions derived in the Supplementary information S1, S2 (box, figure) to calculate the output characteristics of a TMD channel FET in which transport is ballistic — that is, when electrons travel in the channel without being scattered. There are two ways to present the performance of a FET: by the assessment of either transfer characteristics (curve in FIG. 2c; FIG. 3a,b) or output characteristics (curve in FIG. 2g; FIG. 3c). Transfer characteristics are obtained by plotting drain current,  $I_d$ , as a function of the gate voltage (FIG. 3a,b). The on-state current of FETs can be obtained from transfer characteristics at different source–drain voltages. A subthreshold slope (SS; the gate voltage required to increase the drain current by a factor of ten), which indicates the switching speed of the FET, can be extracted from a transfer curve. In our simulated FET consisting of a 2D channel, the switching properties are notable in that the SS is close to  $60 \text{ mV dec}^{-1}$  at 300 K using equation 3 (REF. 16).

$$SS = \left(1 + \frac{C_s}{C_b}\right) \frac{kT}{q} \ln 10 \quad (3)$$

where  $C_s$  is the semiconductor capacitance. The current saturates as a function of the drain voltage because it becomes energetically difficult for the drain electrode to inject carriers into the channel (FIG. 2d,g). At a fixed drain voltage, varying the gate voltage changes the number of electron energy levels that are filled at the injection point of the source, because the band edge states are capacitively coupled to the gate electrode. This changes the current exponentially when  $V_{gs} - V_T \ll V_{th}$ , and as a power



**Figure 2 | Operating principles of a FET with 2D semiconducting materials forming the channels. a** | The cross-section of a typical field-effect transistor (FET) device with source and drain electrodes, a 2D semiconductor channel and a gate electrode that is electrically separated from the channel by an insulator.  $V_{gs}$  is the gate voltage,  $V_{ds}$  is the drain bias and  $V_T$  is the threshold voltage determined by the difference in the work functions of the metal and the semiconductor. **b** | The energy-band diagram in the vertical direction near the source end of the gate, defining various energy scales and band offsets.  $q\phi_B$  is the metal–dielectric barrier height,  $E_{Fs}$  is the quasi-Fermi level of the source electrode,  $qV_i$  is the voltage drop in the insulator,  $E_c$  is the energy of the semiconductor conduction band,  $\Delta E_c$  is the conduction band offset and  $qV_{gs}$  is the barrier modification by the applied gate voltage. **c** | A schematic diagram of the switching transfer characteristics of the FET with the drain biased at saturation, showing how the gate voltage controls the electron population in  $k$  space in the on and off states.  $k_x, k_y$  are the wavevectors that represent the direction of electron wavefunction propagation. **d–f** | The energy-band diagram along the channel from the source to the drain electrode, at various drain voltages as indicated in panel **g**;  $E_{Fd}$  is the quasi-Fermi level of the drain electrode. **g** | Output characteristics of the FET. The electron distribution in the  $k$  space at the source injection point is shown at different drain voltages. The colours indicate the origin of carriers in the channel: red indicates right-moving carriers injected from the source and blue is for left-moving carriers injected from the drain. At a fixed gate voltage, the net density of the right- and left-going carriers does not change, which is why the area of the two half-circles does not change. The change with the gate voltage is indicated in panel **c**. At a large drain voltage shown in panel **d**, it is energetically unfavourable for the drain to inject carriers to the source injection point, which means all carriers at that point are from the source electrode, and the current is saturated beyond this drain voltage (panels **d** and **g**).

law when  $V_{gs} - V_T \gg V_{th}$  (REFS 30,31). When the transistor is switched on, the drain current in a 2D semiconductor FET is  $I_d \sim V_{gs}^{3/2}$ . This is similar to electron transport in a vacuum tube, which is the predecessor of the solid-state switch. This is not a coincidence, because if the transport of electrons is ballistic, the electrons effectively do not see the atoms in their path.

Output characteristics are obtained by plotting  $I_d$  as a function of  $V_{ds}$  for several gate voltages (FIG. 3c). A FET should demonstrate saturation of  $I_d$  above a certain  $V_{ds}$ . This constant current with  $V_{ds}$  is important because, in an integrated circuit, several devices are interconnected and the  $V_{ds}$  being supplied to each device is variable, but the current from each device will be the same because of the saturation behaviour. In digital circuits, the  $I_d$  of a FET has to drive the gates of a few other FETs. This is called fan-out and depends on the saturation of the  $I_d$  of a FET above a certain  $V_{ds}$ . If saturation does not occur,  $V_{ds}$  will oscillate, and the  $I_d$  and the output current of the FET will change. This causes failure of fan-out and reduces the gain of the integrated circuit. The output current of a FET should only depend on the input (gate) voltage and not on the voltage of the integrated circuit. This unidirectionality is at the heart of gain and the practical utility of digital operations of FETs.

The variation of the gate voltage provides useful information about output characteristics of the FET. If the gate voltage is modulated from +0.4 to -0.4 V, that is over 0.8 V, an on/off ratio of  $\sim 10^8$  is possible (FIG. 3a). On-state current densities of  $\sim 1 \text{ mA } \mu\text{m}^{-1}$  (FIG. 3a,b) are achievable at gate and drain voltages of less than 0.5 V. As a result, FETs comprising TMD semiconductors as the channels can be used for energy-efficient, low-power switching applications. Indeed, the performance of TMD FETs is comparable to silicon and III-V semiconductor FETs with shorter channels<sup>11-15</sup>. Moreover, the short-channel degradation is reduced in 2D channels compared with similar down-scaled FET sizes in silicon and III-V semiconductor devices.

If the swing of the gate voltage is reduced to 0.4 V (from +0.2 to -0.2 V), the achievable on/off ratio reduces to  $\sim 10^4$ , at a current of nearly half of that of the on current. This is an endemic problem with FETs; if the input power is constrained by the supply voltage, the switching speed (proportional to the on current) and the probability of successful operation of the electronic task (proportional to the on/off ratio) decrease. With passive gate dielectrics, the best performance is achieved in FETs in which transport is ballistic, because electrons in the channel do not lose energy as a consequence of scattering. This is because ballistic FETs are capable of achieving higher on/off ratios but maintaining high performance under the constraint of low power. One potential way of achieving this is by using a variant of the FET in which transport is not over the barrier but through it, called a tunnelling FET. 2D semiconductors are believed to be highly attractive for use in tunnelling FETs, especially in the scaling limits.

### Influence of effective mass and valleys

Valleys — the local maximum in the valence band or the local minimum in the conduction band — are important

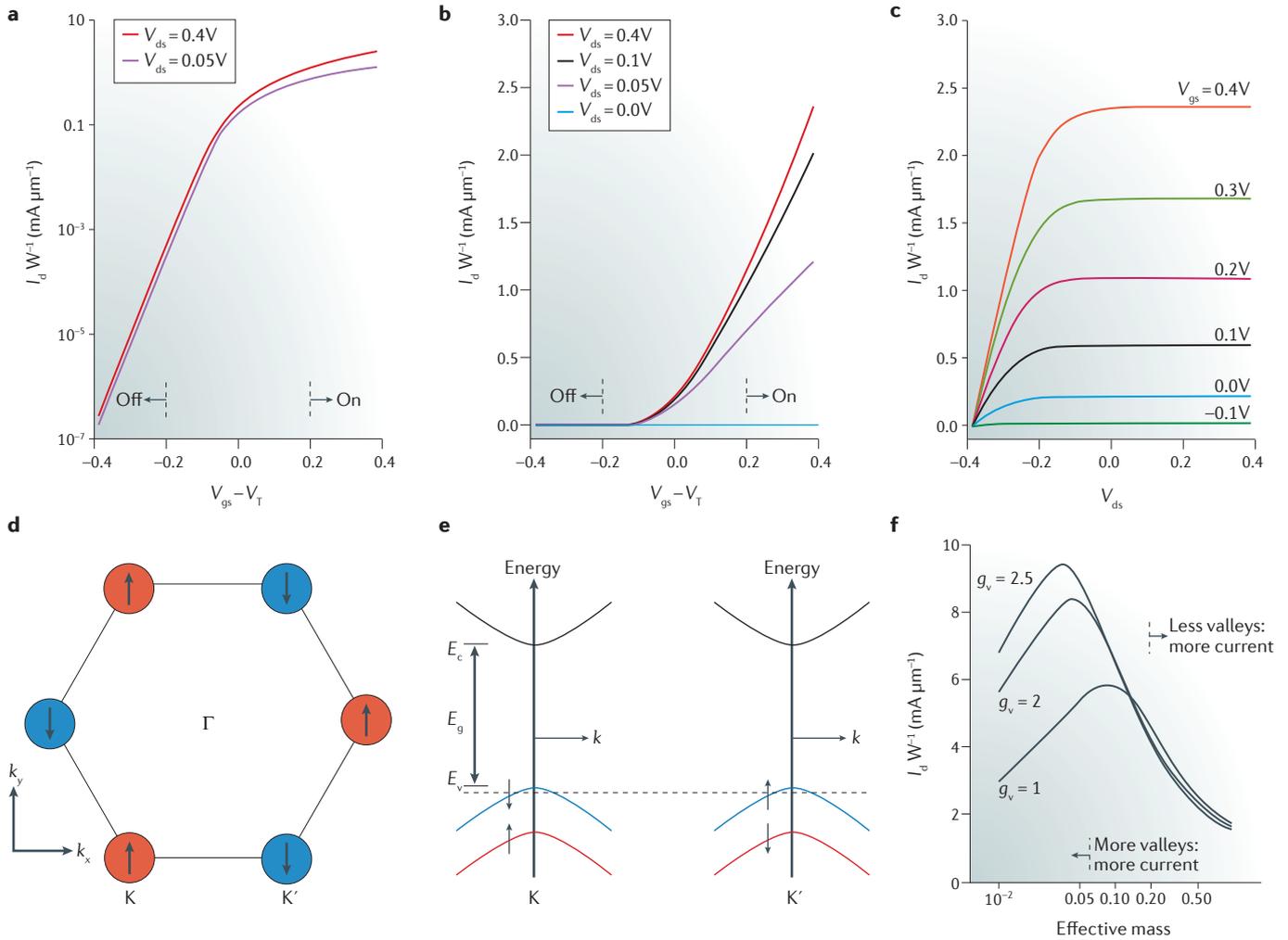
features of 2D TMD semiconductors. Valleys could potentially offer additional degrees of freedom beyond the charge and spin of the electrons used in present day electronics. The first Brillouin zone — a primitive unit cell in the reciprocal lattice (or the  $k$  space) — of  $\text{MoS}_2$  is shown in FIG. 3d. Unlike conventional semiconductors, in some 2D TMD semiconductors there is large (approximately hundreds of millielectron volts) splitting of the valence band because of broken crystal symmetries coupled to a strong spin-orbit coupling<sup>32</sup> (FIG. 3e). The presence of valleys offers opportunities to explore the relative effects of the effective mass of electrons, valley degeneracy and spin degeneracy on the drive current of a ballistic FET. Naïvely, it may be assumed that a small effective mass (that is, a higher velocity) and a large number of valleys will maximize the current. However, the drain current in a ballistic FET is similar to the drive current of FETs with channels composed of III-V semiconductors<sup>33</sup>. The dependence of the on-state current on the effective mass is non-linear (FIG. 3f). More specifically, for effective masses larger than  $\sim 0.2m_0$ , the number of valleys has a small influence on the drive current; however, at smaller effective masses, a higher number of valleys leads to a proportionately higher drive current. The spatial spread of the wavefunction is high for a very small  $z$ -directed effective mass, which is undesirable for the scaling of FETs.

### Contact resistance in FETs

We now consider the realistic impact of contact resistance,  $R_c$ , on FET performance. To reap the benefits of ballistic FETs, the  $R_c$  must be reduced far below the state of the art<sup>8</sup>. The  $R_c$  acts as a severe source-choke — that is, it strongly restricts the injection of electrons into the channel. In addition, the effective gate voltage seen at the source injection point is not  $V_{gs}$  but  $V_{gs} - I_d R_c$ . This reduction in effective voltage leads to degradation in the performance of the transistor, because the current depends very strongly on the effective gate voltage at the source injection point. To appreciate the importance of the  $R_c$ , let us assume that  $R_c \approx 1 \text{ k}\Omega \mu\text{m}$ . To push a current of  $\sim 1 \text{ mA } \mu\text{m}^{-1}$ , the voltage drop across the contacts is  $2I_d R_c \approx 2 \text{ V}$ . The  $R_c$  must be below  $\sim 0.1 \text{ k}\Omega \mu\text{m}$  to reach the ballistic limits in scaled FETs for energy-efficient logic switching. Reduction of  $R_c$  will not just benefit the performance of short-channel ballistic FETs, but also long-channel FETs. This is because the maximum drive current of long-channel FETs is severely limited by the source-choking effect in the presence of a large  $R_c$ . That is, when the channel becomes very long such that there is a significant chance for scattering in the channel, the drain current is written as  $I_d = qn(x)v(x)W$ , where  $n(x)$  is the sheet density of carriers along the channel at point  $x$  and  $v(x)$  is the ensemble velocity of the carriers at that point.

William Shockley introduced the classic model for long-channel FETs by identifying  $qn(x) = C_b(V_{gs} - V_T - V(x))$ , where  $V(x)$  is the local channel potential, and the ensemble velocity may be written as

$$v(x) = \frac{\mu F(x)}{1 + \mu F(x) / v_{sat}} \quad (4)$$



**Figure 3 | Calculated FET characteristics and spin-valley locking in the valence band of 2D TMDs.** The output characteristics of a transition metal dichalcogenide (TMD) ballistic field-effect transistor (FET) with  $m_c^* = 0.5m_0$ ,  $\epsilon_b = 20\epsilon_0$  and  $t_b = 3$  nm are calculated (panels a–c). **a** | Transfer characteristics with the drain current,  $I_d$ , in logarithmic form for two different drain voltages ( $V_{ds}$ ). **b** | Transfer characteristics in linear form at different drain voltages. **c** | Calculated output characteristics of a prototype ballistic 2D semiconductor channel FET at different gate voltages. Note that these calculations do not consider contact resistances. **d** | The electronic structure of a unit cell — referred to as the first Brillouin zone — of monolayer  $\text{MoS}_2$ . The high symmetry points ( $K$ ,  $K'$  and  $\Gamma$ ) of the unit cell in the momentum space are indicated. The  $K$  and  $K'$  points are two inequivalent momentum valleys: the energy maxima and minima. The arrows indicate spins of the valence electrons occupying that valence state.  $k_{x,y}$  are the wavevectors that represent the direction of electron wavefunction propagation. **e** | The valence band splits at these valleys as a consequence of strong spin-orbit coupling. Time reversal symmetry requires spins to be opposite at different valleys. **f** | Drain current as a function of effective mass for three different values of valley degeneracy ( $g_v$ ). The drain currents were obtained for  $V_{gs}$  and  $V_{ds} = 0.4$  V. The cross-over of current scaling with valleys and effective mass depends on the gate capacitance.

where  $\mu$  is the mobility,  $v_{sat}$  is the saturation velocity and  $F(x) = -\partial V(x)/\partial x$  is the local channel field directed along the channel. In the absence of  $R_c$ , the on-state current for a channel length,  $L$ , evaluates to

$$I_d = \mu C_b \frac{W}{L} \left[ (V_{gs} - V_T) V_{ds} - \frac{V_{ds}^2}{2} \right] \left( 1 + \frac{\mu V_{ds}}{v_{sat} L} \right) \quad (5)$$

This expression highlights the importance of mobility, saturation velocity and channel length in long-channel FETs. In this scattering-dominated limit, the details of the band structure, such as valleys and spin, are lumped into the parameters of mobility and the saturation velocity.

The  $R_c$  still has a major role — whenever the voltage drop at the source side is comparable to the applied  $V_{ds}$ , the  $R_c$  causes a severe source-choke.

**Semiconducting 2D materials for FETs**

The demonstration of high performance FETs based on <1-nm thick single-layer  $\text{MoS}_2$  has been an exciting development in the field of 2D electronics<sup>34</sup>. The first single-layer  $\text{MoS}_2$  FETs showed on/off ratios of  $10^8$ , mobility values of  $\sim 200 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  and, importantly, SS values of  $70\text{--}75 \text{ mV dec}^{-1}$  at room temperature<sup>34</sup>. Although these initial mobility values were subsequently revised

downwards<sup>35</sup>, the SS values were close to the theoretical value of  $60 \text{ mV dec}^{-1}$  for standard FET configuration<sup>16</sup>. In addition, the devices showed respectable on-state currents of  $2.5 \mu\text{A } \mu\text{m}^{-1}$ . These device properties were viewed as promising because sharp turn-on, high drive current and high on/off ratios are important parameters for FETs, even for devices with moderate mobility values. These efficient properties reflect the promise of 2D materials for use in low-standby-power integrated circuits.

### Graphene

2D materials can be made by mechanical or chemical exfoliation of bulk layered materials or by chemical vapour deposition of atomically thin layers<sup>36,37</sup>. Graphene is the most widely studied 2D material because it has an extraordinary mobility of  $\sim 25,000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  at room temperature and lacks dangling bonds. Graphene comprises an  $sp^2$ -hybridized carbon-atom arrangement in which three valence electrons form the strong in-plane covalent bonds, and the fourth electron remains in the  $p$  orbital and forms the out-of-plane  $\pi$  bonds. The electrons in the  $p$  orbitals are easily delocalized among the atoms, giving rise to the unique linear dispersion in graphene<sup>38</sup>. Graphene is often referred to as a zero bandgap semiconductor, because carriers in graphene can be influenced by gate voltage in a FET device<sup>39</sup>. However, the absence of a bandgap due to the delocalized electrons in graphene means that it is not possible to switch off a FET comprising graphene as the channel material, which results in impractical on/off ratios of typically  $<10$  (REF. 39). A bandgap of  $\sim 0.1 \text{ eV}$  can be introduced by making narrow ( $\sim 10\text{-nm}$  wide) graphene nanoribbons; however, the fabrication of nanoribbons is challenging. In addition, this approach forms edge states in the 2D structure and thus introduces scattering, resulting in low mobility<sup>40</sup>. Recently, large arrays of  $10\text{-nm}$  graphene nanoribbons have been used to make FETs with a bandgap of  $140 \text{ meV}$ . These FETs show conventional band transport at room temperature and interband tunnelling at low temperatures<sup>41</sup>, which suggest that the edges are smooth. For comparison, it should be stated that world-record-breaking n-type and p-type FETs have been demonstrated with InSb, which has a bandgap of  $170 \text{ meV}$ , mobilities of  $\sim 30,000\text{--}40,000 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  at room temperature and a carrier concentration of  $10^{12} \text{ cm}^{-2}$  (REFS 42,43). Thus, the optimization of the fabrication of graphene nanoribbons could result in FETs with attractive properties.

In recent years, the FET community has shifted its interest to 2D materials with sizable ( $>0.3 \text{ eV}$ ) bandgaps so that low off-state currents and high on-state currents can be realized. More specifically, mono- and few-layered TMDs<sup>34,35,44–61</sup>, phosphorene<sup>62–73</sup> and silicene<sup>74</sup> have been incorporated into FETs.

### 2D transition metal dichalcogenides

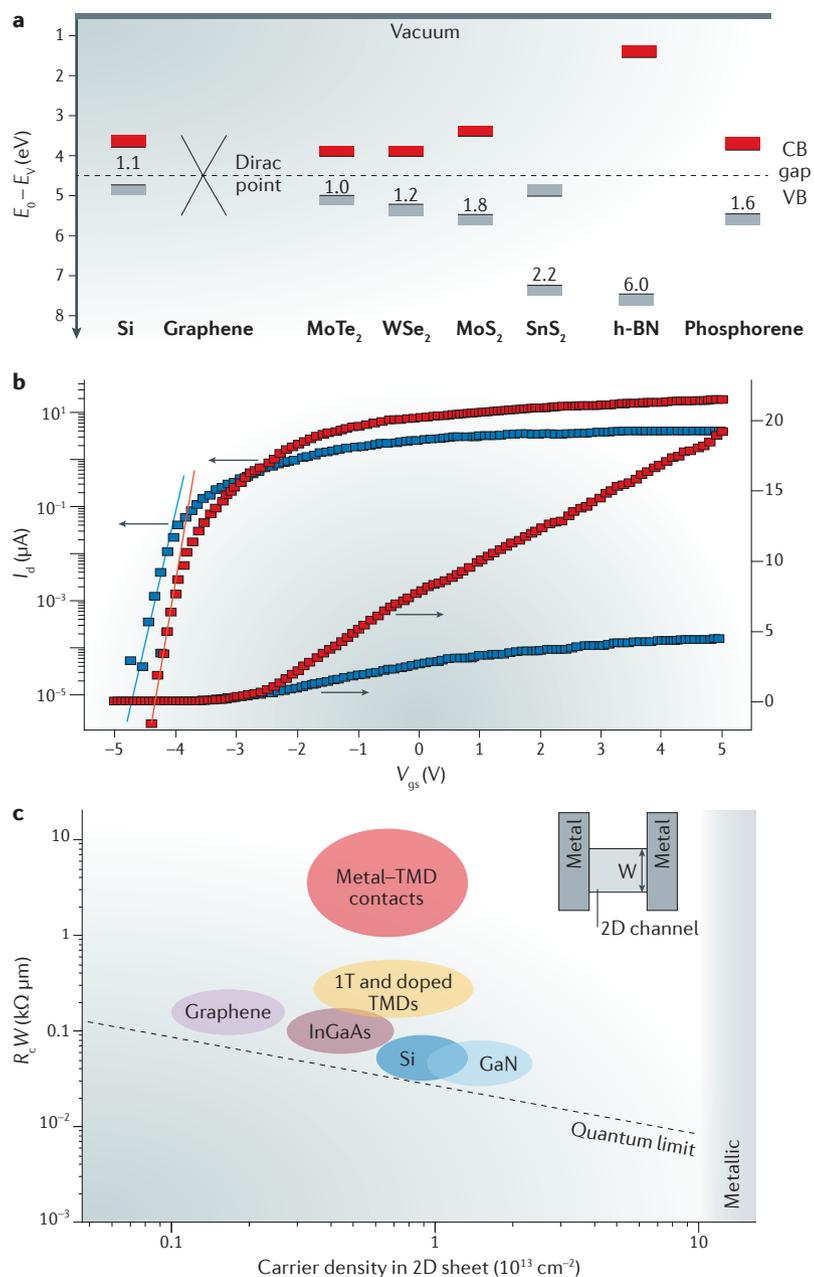
TMDs have the general formula of  $\text{MX}_2$ , where M is a transition metal from group 4, 5 or 6, and X is a chalcogen atom (that is, sulfur, selenium or tellurium). A single layer of these materials consists of three atomic layers in the form of  $\text{X-M-X}$  (REF. 37). Adjacent layers of TMDs are weakly held together by van der Waals forces,

which allow them to be easily exfoliated. There are approximately 40 possible members in the TMD family, many of which have been synthesized. TMDs possess diverse properties: for example,  $\text{MoS}_2$ ,  $\text{WS}_2$  and  $\text{MoSe}_2$  are semiconductors;  $\text{WTe}_2$  and  $\text{TiSe}_2$  are semimetals;  $\text{HfS}_2$  is an insulator; and  $\text{NbS}_2$  and  $\text{VSe}_2$  are true metals. The bandgap of most semiconducting TMDs changes with layer thickness. Bulk layered materials are indirect bandgap semiconductors, whereas single layered TMDs are direct band semiconductors<sup>45,75</sup>. In addition, there is an increase in the value of the bandgap for monolayers compared with the bulk; therefore, it is possible to obtain 2D semiconductors with variable band gaps from  $1.1$  to  $2.2 \text{ eV}$  (REF. 37). However, some TMDs, such as  $\text{ReS}_2$ , are direct bandgap semiconductors in the bulk and in single-layer forms<sup>76</sup>. The energy bands of some important semiconducting TMDs, graphene, h-BN, phosphorene and silicon are shown in FIG. 4a (REF. 16).

The electron orbitals of TMDs are different from those in graphene and h-BN. h-BN has an  $sp^2$ -hybridized atom arrangement and comprises three valence electrons in in-plane  $sp^2$ -hybridized orbitals, like graphene; however, the two remaining electrons of nitrogen form a highly localized lone pair, thus making h-BN a wide bandgap insulator. In contrast to graphene and h-BN, the electronic properties of TMDs are governed by the  $d$  orbitals of the transition metals, and the degree of filling of these orbitals has implications on the electronic structure as well as giving rise to interesting condensed matter phenomena, such as charge density waves, superconductivity, magnetism and hidden states<sup>16,77–79</sup>.

**MoS<sub>2</sub> and other TMD FETs.** Despite the large number of semiconducting TMDs, research on these materials for use in electronic devices has almost exclusively focused on  $\text{MoS}_2$ , although devices with  $\text{WS}_2$ ,  $\text{WSe}_2$  and other TMDs have also been demonstrated<sup>147,50,54–59</sup>.  $\text{MoS}_2$  has attracted this attention because it is readily available (in its natural form), and high quality 2D crystals can be obtained relatively easily.  $\text{MoS}_2$  is also mechanically and chemically robust. In 2007, mechanically exfoliated crystals of  $\text{MoS}_2$  ( $8\text{--}40 \text{ nm}$  in thickness) were used as a channel material in very thin TMD FETs<sup>80</sup>. In a back-gated FET configuration and at a gate voltage of  $-50 \text{ V}$ , n-type transport was observed with mobilities of up to  $50 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  and an on/off ratio of  $>10^5$ . In 2010, top-gated monolayer  $\text{MoS}_2$  transistors with  $\text{HfO}_2$  as the gate dielectric showed mobilities in the range of tens of  $\text{cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  (REFS 34,35), high on/off ratios, high on currents at a source–drain voltage of  $0.5 \text{ V}$  and a low SS of  $74 \text{ mV dec}^{-1}$ . These initial results demonstrated that useful FET properties could be achieved from very thin semiconductor channels and provided encouragement for the community to pursue 2D materials for electronics.

Quantum transport simulations using the non-equilibrium Green's function have been performed to determine the scaling limits of single-layer  $\text{MoS}_2$  transistors with high on/off ratios and good short-channel behaviour, suggesting that  $\text{MoS}_2$  transistors could be suitable for low-power applications<sup>46</sup>. To demonstrate that  $\text{MoS}_2$  transistors are immune to short-channel



**Figure 4 | Energy band alignments and device properties of 2D materials.** **a** | Energy levels of various 2D materials compared with that of silicon. The numbers between the valence band (VB) and conduction band (CB) energies indicate the bandgaps of the materials. The energies with respect to the vacuum level (the work function or the electron affinity) are approximate and subject to experimental refinement. **b,c** | The importance of contact resistance in  $\text{MoS}_2$  field-effect transistors (FETs). **b** | Transfer characteristics of a top-gated device measured at a drain voltage,  $V_{ds}$ , of 1 V. The red curves represent transfer characteristics of phase-engineered low-resistance contacts and the blue curves are for high-resistance contacts. Higher on-state current and lower subthreshold slopes are apparent in the phase-engineered (optimized) device. The linear fits can be used to extract the values of the subthreshold slopes, which are  $95 \text{ mV dec}^{-1}$  for the phase-engineered contacts and  $100 \text{ mV dec}^{-1}$  for the high-resistance contacts. A logarithmic scale of the drain current,  $I_d$ , is shown on the left and a linear scale is on the right. **c** | Contact resistance as a function of 2D-sheet carrier density for various semiconductors against the quantum limit. The use of metallic 1T phase transition metal dichalcogenides (TMDs) as contacts makes it possible to decrease the contact resistance. A schematic illustration of a device is shown in the inset. Panel **a** is adapted with permission from REF. 16, IEEE. Panel **b** is from REF. 60, Nature Publishing Group. Panel **c** is from REF. 88, Nature Publishing Group.

effects, FETs with channel lengths of up to 100 nm have been fabricated<sup>81</sup>. In addition, the performance limit of these  $\text{MoS}_2$  transistors is a consequence of the high  $R_c$  between the contacts and the  $\text{MoS}_2$  channel, and as such, contacts that allow full transmission of electrons from the electrodes to the channel are required for the realization of high performance short-channel devices. In a further study, the performance metrics of 5-nm-channel FETs were shown to be comparable to the ITRS 2026 low operating power technology requirements<sup>82</sup>.  $\text{MoS}_2$  transistors have been fabricated with a range of metals as the contact electrodes, and it has been shown that Fermi-level pinning at the conduction band of  $\text{MoS}_2$  strongly influences the metal/ $\text{MoS}_2$  interface<sup>49</sup>. The best device performance was achieved for scandium contacts owing to high carrier injection and a low  $R_c$  of  $0.65 \text{ k}\Omega \mu\text{m}$  (REF. 49). Another interesting approach for ensuring a low  $R_c$ , and hence high performance devices, is to degenerately dope the contacts of  $\text{MoS}_2$  with elements, such as potassium<sup>52</sup>. However, doping with materials or chemicals that readily react with the environment or that evolve just above room temperature introduces instability and can cause the properties of the device to deteriorate over time. In addition to single-layer  $\text{MoS}_2$  FETs, interesting characteristics — including mobilities of  $\sim 100 \text{ cm}^2 \text{V}^{-1} \text{s}^{-1}$  — can be obtained from multilayer  $\text{MoS}_2$  FETs<sup>83</sup>. Multilayer  $\text{MoS}_2$  has other attractive features, such as high current modulation, low SS and the ease of growing multilayer  $\text{MoS}_2$  over a large area.

**Challenges for  $\text{MoS}_2$  FETs.**  $\text{MoS}_2$  FETs exhibit promising results; however, fundamental and practical challenges remain. For example, although the effective masses of the conduction and valence band edges have been calculated to be approximately symmetric (that is, the electron effective mass is  $\sim 0.57$  and the hole effective mass is  $\sim 0.66$ ),  $\text{MoS}_2$  FETs usually exhibit n-type characteristics, suggesting that the channel material has been unintentionally doped<sup>34</sup>. The exact origin of the n-type behaviour is unclear, but the presence of impurities, such as rhenium and gold, sulfur vacancies and Fermi-level pinning near the conduction band, are possible causes<sup>84,85</sup>. It is also important to note that 2D  $\text{MoS}_2$  is non-stoichiometric with a variable Mo/S ratio from  $\sim 1:1.8$  to  $1:2.3$  (REF. 85) and that the energy of the Fermi level can vary over the surface. The structural defects, such as sulfur vacancies<sup>86</sup>, along with impurities can limit the performance of 2D  $\text{MoS}_2$  FETs. Synthesis of high purity and stoichiometric  $\text{MoS}_2$  should allow the realization of ambipolar FETs. Furthermore, development of controlled doping techniques is required for tuning the polarity of the devices. These limitations can be partially overcome by sandwiching  $\text{MoS}_2$  between layers of h-BN, which reduces the effect of charged impurities on transport properties and allows mobilities that are close to what is theoretically possible ( $\sim 30,000 \text{ cm}^2 \text{V}^{-1} \text{s}^{-1}$ ) to be obtained, although at low temperatures<sup>26</sup>. Other TMDs — for example,  $\text{WSe}_2$  — exhibit primarily p-type characteristics<sup>47</sup>, and  $\text{WS}_2$ , which tends to be more stoichiometric than  $\text{MoS}_2$ , exhibits ambipolar behaviour<sup>87</sup>.

**Improving contact resistance.** The resistance of contact electrodes is a major limitation in short-channel FETs comprising silicon and III–IV semiconductors. It has also limited the performance of FETs incorporating TMD or other 2D materials as the channel material<sup>188,89</sup>. Resistance at the source and drain contacts is around  $10\text{ k}\Omega\mu\text{m}$  in  $\text{MoS}_2$ -based FETs<sup>49,60</sup> — over 100 times higher than that of silicon-based electronics<sup>8</sup> ( $<20\text{ }\Omega\mu\text{m}$ ). Contacts in silicon or III–IV electronics are degenerately doped or regrown to achieve exceptionally low resistance and sharp interfaces<sup>15</sup>. These methodologies pose challenges in 2D materials because of their atomically thin nature. Future FETs will require single-layer channels and multi-layer heavily doped contacts. In multilayered  $\text{MoS}_2$  FETs, scandium electrodes<sup>49</sup> and doping by physisorbed molecules<sup>52</sup> have been successfully implemented to reduce the  $R_c$ . In another study, graphene has been shown to be an effective contact electrode material for ultrathin  $\text{MoS}_2$  FETs<sup>26,90,91</sup>. Phase engineering has also been used to decrease the  $R_c$  in 2D  $\text{MoS}_2$  devices down to  $200\text{ }\Omega\mu\text{m}$  (REF. 60). More specifically, TMDs can exist in either the semiconducting 2H phase or the metallic 1T phase, and interfacing these two phases in a controllable manner results in atomically sharp metal/semiconductor junctions<sup>92</sup>. The decrease in  $R_c$  using phase engineering allows current capacities to exceed  $100\text{ }\mu\text{A}\mu\text{m}^{-1}$  in TMD FETs<sup>60</sup> (FIG. 4c). Recently, phase engineering in  $\text{MoTe}_2$  by laser irradiation has been shown to fabricate Ohmic contacts<sup>93</sup>. Phase engineering, therefore, offers a route for building lateral heterostructures with low  $R_c$ , but additional work is necessary to achieve values comparable to state-of-the-art materials and to approach the quantum limit<sup>88,89</sup> (FIG. 4d). In addition, methods that are compatible with electronic processing and have sufficiently good spatial resolution for device patterning are required.

Calculations have revealed that the effective masses of electrons and holes pose fundamental challenges for achieving high switching speeds in  $\text{MoS}_2$  and other TMD-based FETs<sup>46</sup>. However, the large effective masses of TMDs are beneficial for switching devices, such as tunnelling FETs. Nevertheless, the excellent performance of FETs outlined above offers opportunities for realizing large area, transparent and flexible electronics<sup>90,91,94–98</sup>. Recent progress on the wafer-scale synthesis of high quality  $\text{MoS}_2$  suggests that large-area monolayer  $\text{MoS}_2$ -based FETs<sup>99</sup> could find use in electronics for which exceptionally high processing speeds are not required.

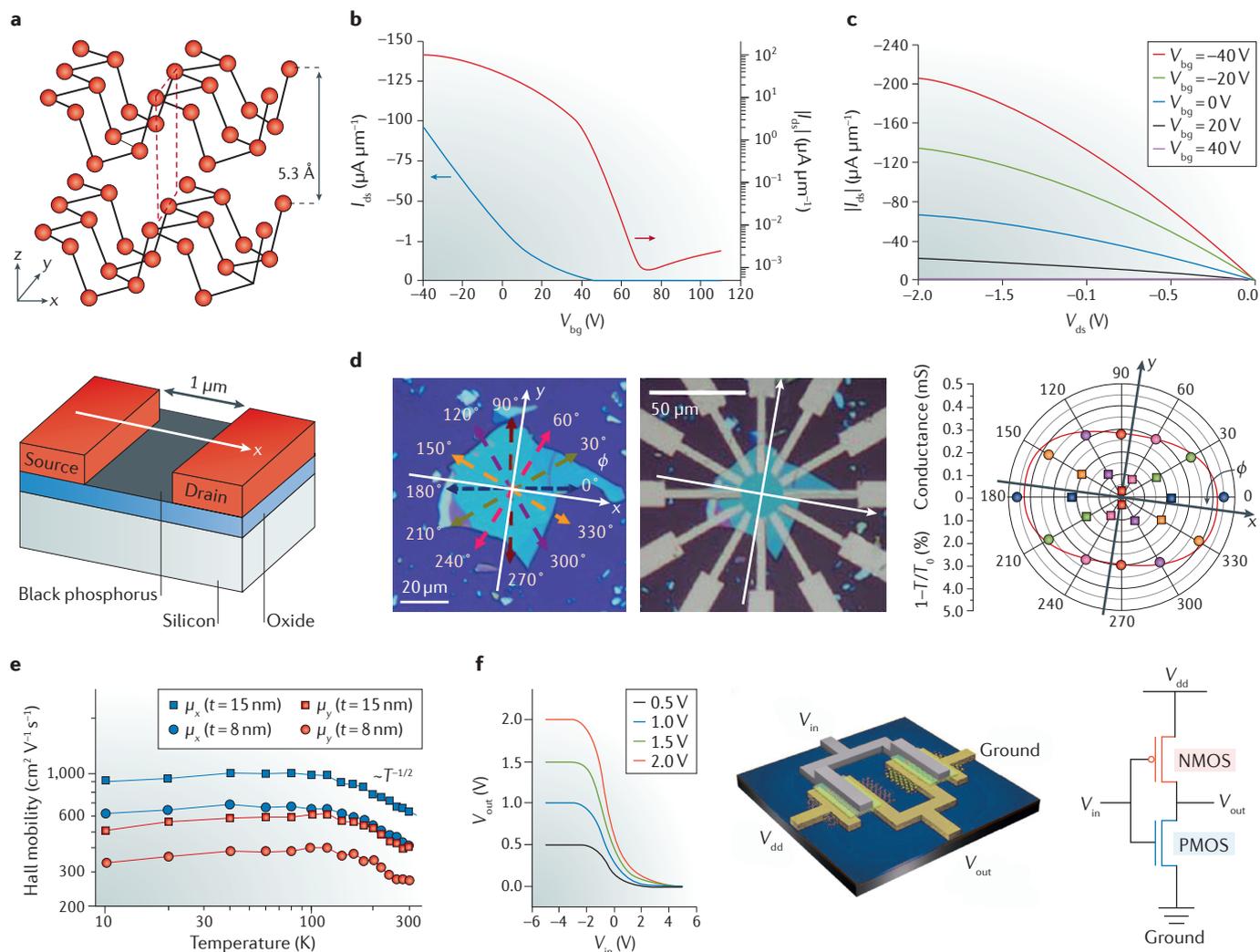
### Phosphorene FETs

In 1914, black phosphorus was synthesized by compressing red phosphorus under high pressure and at high temperatures<sup>100</sup>. Black phosphorus is a layered material with covalent bonds in the layers and weak, van der Waals bonds between the layers<sup>62,101</sup> (FIG. 5). The layers have a puckered honeycomb structure that, if stretched laterally, would result in a graphene-like hexagonal structure. Each layer consists of phosphorus atoms with three valence  $p$ -orbital electrons that form three covalent bonds to each other. Because all valence electrons are involved in covalent bonding, black phosphorus is a semiconducting material with a bandgap of  $\sim 0.3\text{ eV}$  (REF. 102). It typically

behaves as a  $p$ -type semiconductor, but the origin of this behaviour is unclear, although the polarity of the FETs can be tuned by selecting contact metals with appropriate work functions. The weak bonding between each layer allows the material to be exfoliated down to a few layers with ease<sup>62–73</sup>. The bandgap varies as a function of the number of layers and reaches a value of  $\sim 2\text{ eV}$  for single-layer black phosphorus<sup>103</sup>. Single- and few-layered black phosphorus is referred to as phosphorene. The bulk electronic properties of black phosphorus are intriguing. In particular, electron and hole mobilities of  $\sim 1,000\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$  have been observed<sup>62,66</sup>. At low temperatures, mobilities as high as  $15,000\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$  for electrons and up to  $50,000\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$  for holes have been predicted<sup>104</sup>. This has motivated research in electronic properties of 2D phosphorene for applications in photodetectors<sup>105</sup> and solar cells<sup>65</sup>, as well as in FETs<sup>62–68</sup>.

Single-layer phosphorene FET devices have not been widely reported because there has not been a pressing need to isolate phosphorene monolayers for this application. This is because although the value of the bandgap varies with the number of layers, phosphorene remains a direct bandgap semiconductor. The thickness of few-layered phosphorene in electronic devices is around  $10\text{ nm}$  (REF. 106). Isolation and eventual growth of stable single- or few-layered phosphorene is essential for the development of high performance, scalable FETs. Typical transfer and output characteristics of back-gated phosphorene FETs are shown in FIG. 5b,c (REF. 66). Most reported FETs incorporating phosphorene have been in the back-gated configuration; these devices yield mobilities ranging from  $100$  to  $1,000\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$  with on/off ratios of  $\sim 10^4$  and a SS of  $1\text{ V dec}^{-1}$  (REFS 62–68). Top-gated devices improve the SS values to  $\sim 100\text{ mV dec}^{-1}$  (REF. 107). The drive current capacity of phosphorene is  $\sim 100\text{ }\mu\text{A}\mu\text{m}^{-1}$  at a drain bias of  $0.5\text{ V}$ , which is comparable to that achieved in TMDs. Saturation in the output characteristics, similar to the output curves of  $\text{MoS}_2$  FETs, is also observed in phosphorene FETs. A Hall mobility,  $\mu$ , reaching  $6,000\text{ cm}^2\text{ V}^{-1}\text{ s}^{-1}$  has been achieved in few-layer phosphorene, which has enabled the observation of the quantum Hall effect at temperatures of  $<10\text{ K}$  (REF. 108).

An interesting feature of phosphorene is that it is electronically anisotropic; that is, transport in the armchair direction ( $x$  direction in FIG. 5a) is much more efficient than in the zigzag direction ( $y$  direction in FIG. 5a). Polarization-resolved infrared spectroscopy and angle-resolved direct current conductance measurements have been performed to elucidate the degree of anisotropy in phosphorene (REFS 62,66). Using the directional dependence of low-field conductivity for an anisotropic material given by  $\sigma_\theta = \sigma_x \cos^2(\theta - \varphi) + \sigma_y \sin^2(\theta - \varphi)$  to calculate the conductance, an excellent fit with the experimentally measured data that is observed (FIG. 5d). In the equation,  $\sigma_x$  and  $\sigma_y$  are the conductivities in the  $x$  and  $y$  directions in the crystal (FIG. 4a),  $\theta$  is the angle with respect to  $0^\circ$  along which the conductance is measured,  $\sigma_\theta$  is the conductivity in the  $\theta$  direction and, finally,  $\varphi$  is the angle between the  $x$  direction and the  $0^\circ$  reference. It has been reported that  $\varphi = -8^\circ$ , which is in agreement with results using polarized infrared spectroscopy<sup>66</sup>. The high and low conductivity



**Figure 5 | 2D phosphorene for FETs.** **a** | The orthorhombic crystal structure of black phosphorus (top) from which few-layered phosphorene is obtained. Each phosphorus atom is bonded to three other phosphorus atoms to form a six-membered ring. The individual puckered sheets are linked by weak van der Waals bonding with a spacing of 5.3 Å. A schematic illustration of the back-gated field-effect transistor (FET) is depicted below. The  $x$  direction in the FET device is correlated with the  $x$  direction of the phosphorene lattice. **b** | Drain–source current,  $I_{ds}$ , as a function of back gate bias,  $V_{bg}$ , transfer curves for a 5-nm thick phosphorene channel of length  $L = 1 \mu\text{m}$ , showing p-type behaviour. **c** |  $I_{ds}$  versus drain-source voltage ( $V_{ds}$ ) output characteristics showing current saturation. **d** | An optical micrograph of a 30-nm black phosphorus flake (left); black phosphorus flake with 12 electrodes spaced  $30^\circ$  apart (middle); and direct-current conductivity and relative infrared extinction (right) measured along the six directions shown in the image on the left. The circles represent direct current conductance and the squares represent polarization-resolved extinction at  $2,700 \text{ cm}^{-1}$ . The colours of the dots and squares correspond to the colours in the image on the left. **e** | Angle-resolved Hall mobility,  $\mu_{x,y}$ , measurements for phosphorene films show that thicker films have higher mobilities. Also, mobility along the  $x$  direction is  $\sim 1.8$  times higher than in the  $y$  direction. The hole carrier concentration is constant at  $6.7 \times 10^{12} \text{ cm}^{-2}$ . **f** | The voltage transfer curve of an inverter device consisting of n-type metal-oxide semiconductor (NMOS)  $\text{MoS}_2$  FET and few-layer phosphorene p-type metal-oxide semiconductor (PMOS) FET (left). A schematic diagram of the inverter device is shown in the middle, and a circuit representing the inverter device is depicted on the right.  $V_{dd}$ , power supply voltage;  $V_{in}$ , input voltage;  $V_{out}$ , output voltage. Panels **a–e** are from REF. 66, Nature Publishing Group. Panel **f** (left) is courtesy of P. Ye, Purdue University, USA. Panel **f** (middle and right) is published with permission from REF. 62, American Chemical Society.

directions also represent the high and low mobility directions, and the ratio of  $\sigma_x/\sigma_y$  can be extracted to be  $\sim 1.5$ , which is slightly smaller than the experimentally measured Hall mobility values (FIG. 5e). Hall mobility measurements in the  $x$  and  $y$  directions yield a  $\mu_x/\mu_y$  ratio of  $\sim 1.8$ , because the calculations do not consider the spreading of current in the material<sup>66</sup>.

A CMOS logic circuit has been demonstrated that consists of top-gated p-type phosphorene and few-layered n-type  $\text{MoS}_2$  FETs<sup>69</sup> (FIG. 5f). Voltage transfer characteristics of these FETs show a clear transition to zero within the input voltage range from  $-5$  to  $+5$  V. These results also highlight the integration of two heterogeneous 2D materials for FETs.

A principal challenge associated with phosphorene is its environmental stability<sup>109–114</sup>. The surface of phosphorene is hydrophilic as a consequence of a permanent out-of-plane dipole moment and, as a result, it readily oxidizes<sup>111</sup>. More specifically, it has been recently elucidated that a combination of oxygen, light and moisture leads to rapid degradation of the material<sup>110</sup>. Some progress has been made on capping phosphorene using h-BN<sup>112</sup>, atomic layer deposited  $\text{Al}_2\text{O}_3$  (REFS 113, 114) and a double-layer comprising  $\text{Al}_2\text{O}_3$  and a hydrophobic fluoropolymer<sup>110</sup>. The latter double-layer approach appears to offer indefinite stability by ensuring conformal sidewall coverage to prevent diffusion from edges and moisture resistance from an effective hydrophobic surface.

Phosphorene also suffers from a high  $R_c$ , thus limiting the overall performance of the FETs. Using Ti–Au contacts, Schottky barrier heights of around 0.2 eV have been measured. New graphene side contacts have been achieved but the overall  $R_c$  remains in the  $\text{k}\Omega\text{-}\mu\text{m}$

range<sup>112</sup>. However, unlike  $\text{MoS}_2$ , the Fermi level is not pinned and both p- and n-type devices can be achieved. This also suggests that phosphorene has good structural integrity and a low defect concentration, as long as the material is not exposed to the environment.

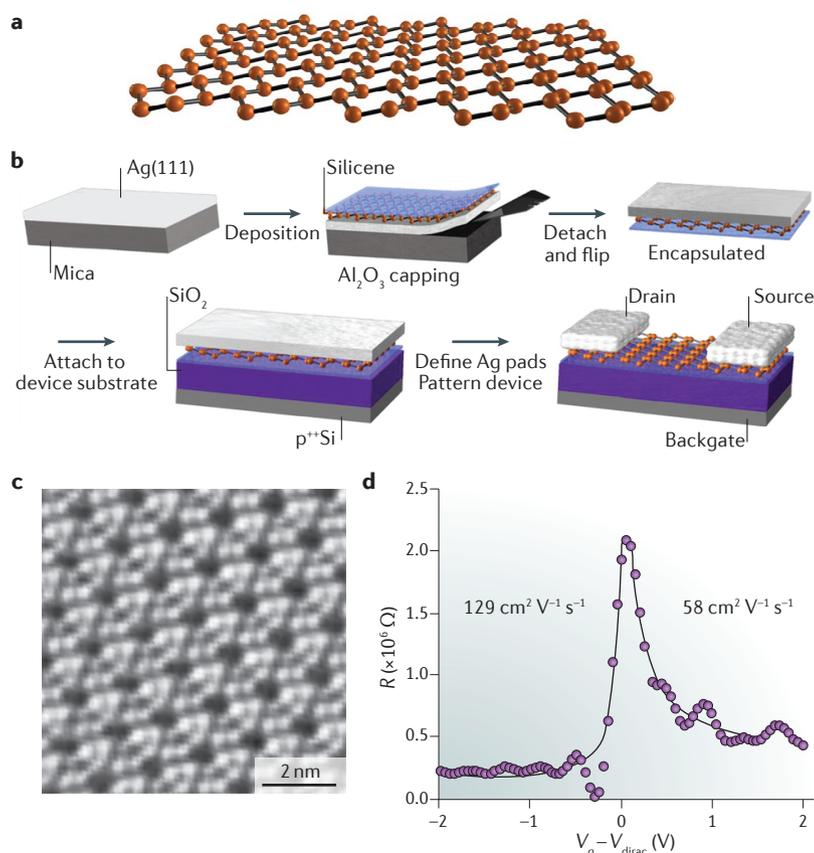
### Silicene FETs

Silicene is an allotrope of silicon with a buckled, six-membered ring 2D structure that is analogous to graphene<sup>115</sup> (FIG. 6a). It is, like phosphorene, highly unstable under ambient conditions because of mixed  $sp^2$ – $sp^3$  hybridized bonding<sup>116</sup>. The use of silicene in electronic devices is particularly exciting because, if high performance devices can be realized, the existing CMOS infrastructure that is based almost entirely on the processing of silicon could be easily applied.

Silicene deposited under ultrahigh vacuum conditions on bulk single-crystal Ag(111) or thin-film substrates has been reported (FIG. 6c). An effective encapsulation and transfer process (FIG. 6b) has allowed the fabrication of silicene-based FETs, which have good mobilities at room temperature<sup>74</sup> (FIG. 6d). More fundamentally, the density of states in silicene is predicted to be tuned between gapped (semiconducting) and gapless (metallic) to gapped states by applying a vertical electric field. The gapless version of silicene is predicted to host topologically non-trivial states, which is why its physics, in some sense, is richer than that of graphene or  $sp^3$ -hybridized silicon. The reported transfer characteristics of silicene FETs are similar to those obtained for graphene with an on/off ratio of  $\sim 10$ , and hole and electron mobilities of  $\sim 129 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$  and  $\sim 58 \text{ cm}^2 \text{ V}^{-1} \text{ s}^{-1}$ , respectively, using the ambipolar diffusive transport model<sup>74</sup>. The intrinsic carrier concentration in silicene ( $\sim 5 \times 10^9 \text{ cm}^{-2}$ ) is believed to be an order of magnitude less than that in graphene, although the Fermi velocity is comparable<sup>117,118</sup>. These factors suggest that silicene has a small but finite bandgap of  $\sim 210 \text{ meV}$  (REF. 74). The room-temperature oscillatory conductivity behaviour reported for silicene (FIG. 6d) could potentially point towards rich transport physics, which is sure to be explored in the future.

### Tunnelling FETs based on 2D Materials

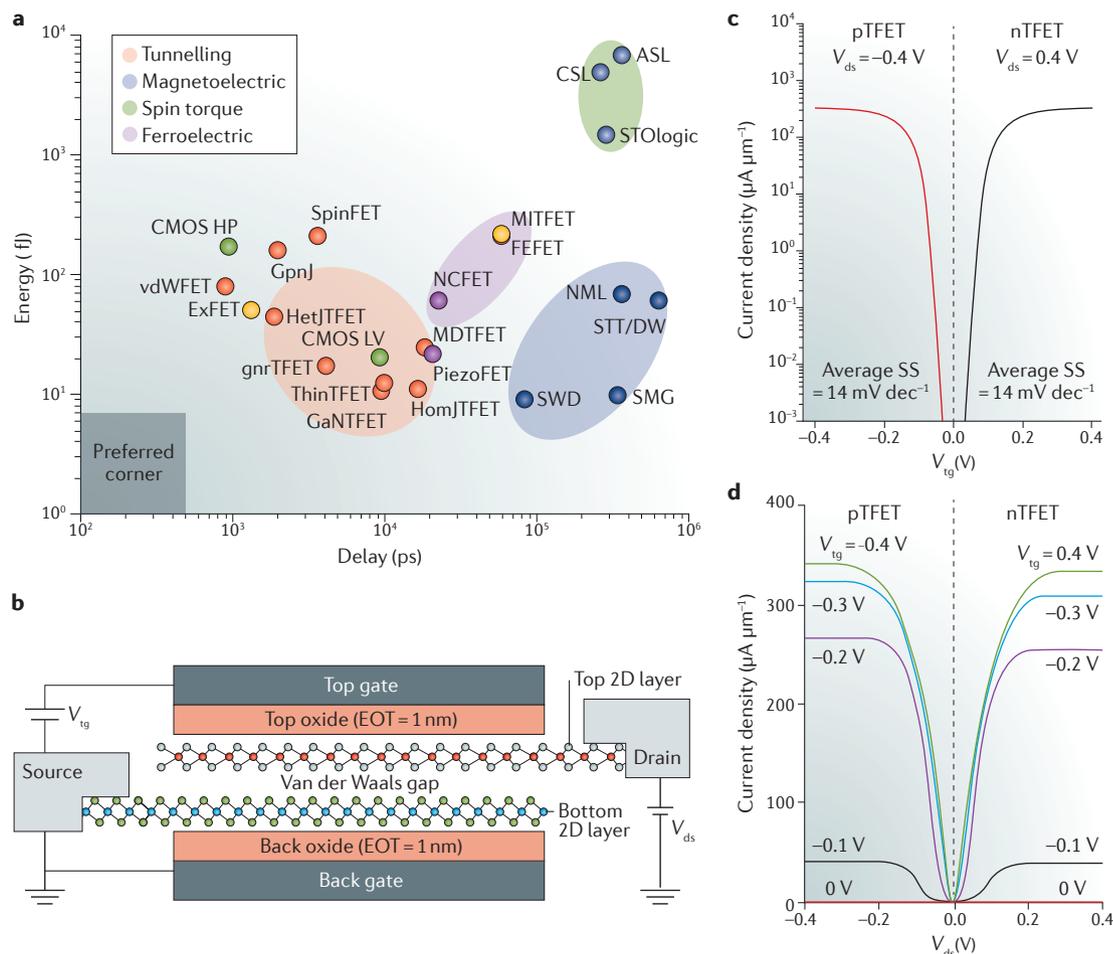
In the discussion of ballistic and long-channel FETs, constraints on the voltage swing were shown to severely restrict the on/off ratio and the current drive. The combined metric for such Boolean (on/off-based) logic operations is termed the energy–delay product, which must be minimized. The energy–delay product to complete the same circuit-level operation for several electronic and magnetic logic devices are shown in FIG. 7a (REF. 119). Some of these devices (for example, CMOS) are common, some have been experimentally demonstrated and others have only been proposed. The delay in a charge-based device is of the form  $\tau = CV/I$ , where  $I$  is the current drive and  $Q = CV$  is the net charge to be switched. The most desirable situation (or preferred corner shown in FIG. 7a) is for FETs to perform very fast computation (lowest delay) with the least amount of energy. We see that a high-performance CMOS is fast but takes more energy to perform logic operations; these devices form



**Figure 6 | 2D silicene FETs.** **a** | A schematic illustration of silicene showing the buckled six-membered ring structure. **b** | A fabrication route to a silicene back-gated field-effect transistor (FET) showing epitaxial growth of silicene on an Ag(111) crystalline film, *in situ* capping with  $\text{Al}_2\text{O}_3$ , delamination and encapsulation of the film, and the use of the native Ag film to produce the contact electrodes. **c** | Scanning tunnelling microscope image of Si overlayers with  $4 \times 4$  superstructures. **d** | A transfer curve plotted of response ( $R = V_d/I_d$ , where  $V_d$  is the drain voltage and  $I_d$  is the drain current) as a function of overdrive voltage ( $V_g - V_{\text{dirac}}$ , where  $V_g$  is the gate voltage and  $V_{\text{dirac}}$  is the voltage at which  $R_{\text{max}}$  occurs). The mobility values for electrons and holes are indicated. The device indicates the range of electron and hole mobility values that can be obtained from silicene FETs. The devices fabricated<sup>74</sup> indicate that the bandgap in silicene is  $\sim 0.2 \text{ eV}$ . Panels **a–d** are from REF. 74, Nature Publishing Group.

the core of powerful microprocessors in electronics today. By contrast, a low-voltage CMOS consumes less energy (by an order of magnitude), but is slower and has enabled the recent explosion of computational devices for portable electronics, such as smart phones and tablets. These slower devices lack the number-crunching capability of desktop and notebook computers but can be operated using battery power for at least 24 hours. A major challenge in semiconductor device design today is to build a device that gets closer to the preferred corner (FIG. 7) of a low-energy and low-delay product.

In a traditional metal oxide semiconductor FET, thermal electrons in a semiconductor band go over a gate-controlled barrier and, as a result, there is a tail of high-energy electrons that do not see the barrier and trickle from the source to the drain electrode<sup>30,31</sup>. This gives rise to a leakage current in the off state of the device. The electrostatics that drive the flow of charges in a FET give a SS close to  $\sim 60 \text{ mV dec}^{-1}$  (as calculated in FIG. 3a and described in FIG. 8a). To reach the desired energy–delay product, the device switch needs to be made with a steeper SS (FIG. 8b). This can be achieved by changing



**Figure 7 | Future FET technologies.** **a** | Intel’s benchmarking of future field-effect transistor (FET) technologies comparing complementary metal-oxide semiconductor (CMOS) high-performance and CMOS low-voltage in the energy–delay product metric for digital logic circuits. 2D semiconductor-based tunnelling FETs, such as the graphene nanoribbon tunnelling FET and the thin-tunnelling FET, show significant promise. **b** | A schematic illustration of the thin-tunnelling FET projected performance of the device. The device structure consists of vertically stacked p- and n-type transition metal dichalcogenides (TMDs) connected to drain and source electrodes, respectively.  $V_{tg}$  is the gate voltage in a tunnelling FET and  $V_{ds}$  is the drain voltage. **c** | The transfer properties reveal exceptionally low subthreshold slopes and high on/off ratios. **d** | The output characteristics exhibit excellent ambipolar modulation with tunnel gate voltage as well as current saturation. ASL, all spin logic; CMOS HP, high-performance silicon CMOS; CMOS LV, low-voltage silicon CMOS; CSL, charge spin logic; EOT, equivalent oxide thickness; ExFET, excitonic FET; FEFET, ferroelectric FET; GaNTFET, gallium nitride tunnelling FET; gnrTFET, graphene nanoribbon TFET; GpnJ, graphene p–n junction; HetJTFET, heterojunction III–IV TFET; HomJTFET, Homojunction III–IV TFET; MITFET, metal–insulator transistor FET; NCFET, negative capacitance FET; NML, nanomagnetic logic; PiezoFET, piezoelectric FET; SMG, spin majority gate; SpinFET, spin FET (Sughara–Tanaka); STO, spin torque oscillator; STT/DW, spin torque domain wall; SWD, spin wave device; ThinTFET, 2D heterojunction interlayer TFET; vdW FET, van der Waals solids (or 2D Materials) FET. Panel **a** is adapted with permission from REF. 119, IEEE. Panels **b–d** are adapted with permission from REF. 120, IEEE.

from a passive gate insulator to an active one, for example, by using a ferroelectric layer that could boost the gate potential and possibly enable a steeper SS<sup>120,121</sup>.

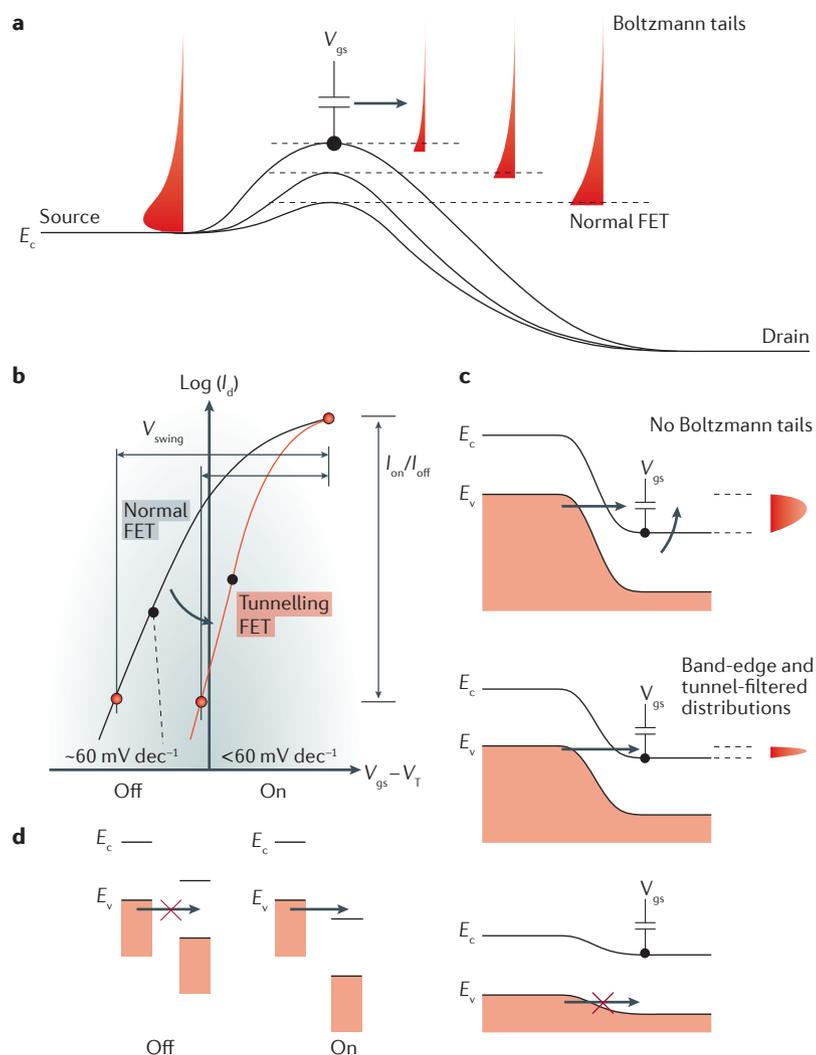
Another method to achieve a steep SS is to remove the high-energy tail that leaks from the source to the drain electrode in a normal FET. This ‘energy filtering’ can be achieved by introducing an energy gap at the appropriate

energies in the density of states of the source of the FET. For example, inserting superlattices in 1D materials to break up the continuous energy band into a set of discrete minibands with gaps can lead to steep subthreshold FETs<sup>122</sup>. An easier way is if the source of the electrons is from the top of the valence band, then the valence band edge and the gap above it are a natural energy filter (FIG. 8c). However, for electrons to flow, they must overcome the barrier of the energy gap. Electrons have a wave-like nature that allows them to use evanescent states to tunnel through the gap at a low-energy cost. Transistors based on interband tunnelling, termed tunnelling FETs, are expected to enable low-power transistor switches that beat the SS limit of  $60 \text{ mV dec}^{-1}$  (REF. 123) (FIG. 8b). Getting below the traditional SS value has not been difficult; semiconductor tunnelling FETs that show steeper than  $60 \text{ mV dec}^{-1}$  switching have been demonstrated using silicon, germanium and carbon nanotubes<sup>124</sup>. The current drives in many such tunnelling FETs have been extremely low, leading to long delays,  $CV/I$ , making them less attractive than, for example, the low-voltage CMOS in FIG. 7a. The interband tunnelling current decreases as an exponential function of the bandgap. Narrow-band-gap semiconductors allow larger tunnelling currents, but because they are scaled down to nanoscale dimensions, the bandgaps in thinned 3D materials increase owing to strong quantum confinement effects<sup>16</sup>.

In this field of tunnelling FETs, crystalline 2D semiconductors are expected to provide a unique solution. Although the bandgap of a single-layer 2D crystal semiconductor is well defined, by choosing a suitable heterojunction of two layers — one n-type layer and one p-type layer (FIG. 7b) — it is possible to maximize the interlayer tunnelling current but maintain a very low SS. The tunnelling current can be boosted by the proper choice of heterojunction band offsets between the two 2D crystal layers. For example, in III–V heterostructures, the tunnelling current has been significantly increased by the choice of type-II (staggered) or type-III (broken) band alignments<sup>125</sup> (FIG. 8d). Similar heterostructure design opportunities exist for 2D crystal interlayer tunnelling FETs. This type of tunnelling transistor could be the thinnest possible manifestation of a tunnelling FET, which can also be scaled to the smallest dimensions, far below what may be feasible with traditional 3D semiconductors. The race is currently on to realize this proposed device, called the thin-tunnelling FET. A first step in this direction is the demonstration of interlayer tunnelling in layered 2D crystal gapped semiconductor heterostructures<sup>126–128</sup>. A tunnelling FET with a 2D  $\text{MoS}_2$  channel contacted with highly doped germanium source electrodes has been reported with a SS of  $\sim 32 \text{ mV dec}^{-1}$  and a supply voltage of  $< 0.1 \text{ V}$ , although at low current values<sup>129</sup>. The goal of achieving such tunnelling FETs is also forcing controlled growth, doping and heterostructure design, and will open several new opportunities in the future of 2D crystal electronics.

## Conclusions

The inherent advantages of 2D materials — for example, the absence of dangling bonds and excellent gate



**Figure 8 | Operating mechanism of tunnelling FETs.** **a** | An energy band diagram showing the Boltzmann tail of electrons in a semiconductor band that leads to the  $60 \text{ mV dec}^{-1}$  subthreshold slope (SS) and drain current,  $I_d$ , leakage in a normal field-effect transistor (FET). The barrier for injection from the source can be lowered by the application of a gate voltage,  $V_{gs}$ . **b** | The transfer characteristics (drain current as a function of gate voltage of a FET) of a normal FET and a tunnelling FET. The tunnelling FET can switch at a SS of  $< 60 \text{ mV dec}^{-1}$  that can reduce the voltage swing ( $V_{\text{swing}}$ ) — the voltage difference between the off and on states — but maintain the same on/off ratio. **c** | If the source is replaced by a filled band, the Boltzmann tail is cut off by the energy gap (top). Interband Zener tunnelling and the density-of-states overlap of the filter electrons of the valence and conduction bands in a narrow energy window, enabling a sharper turn-off, are depicted in the middle and bottom. **d** | The on current can be significantly boosted by choosing two 2D crystals with type-II or type-III heterostructure band alignment, and by interlayer tunnelling of carriers between them. The tunnelling FET can potentially allow for energy-efficient switching, because it reduces the voltage swing necessary to achieve high on current by virtue of the steep SS.  $E_c$ , energy of the conduction band;  $E_v$ , energy of the valence band;  $I_{\text{off}}$ , off-state current;  $I_{\text{on}}$ , on-state current;  $V_T$ , threshold voltage.

electrostatics — make them promising candidates for high performance electronics. However, the implementation of 2D semiconductors in electronic devices remains in its infancy, and some basic techniques of modern electronics are yet to be developed.

Research on 2D materials for electronics has largely focused on achieving high carrier mobilities, and this parameter is often used as a figure of merit for evaluating potential materials for FETs. However, parameters such as low SS across a wide range of gate biases, resulting in large on/off ratios at low input voltages, are more relevant for addressing the challenges of the field of electronics. In addition, contact resistance is a major factor limiting performance of 2D semiconductor FETs, especially at short-channel lengths, and must be lowered by at least an order of magnitude to  $<100\ \Omega\ \mu\text{m}$ . This requires the contact regions to be degenerately doped by either the direct growth of metallic multilayer 2D materials or implantation. In 2D TMDs, using metallic 2D phases is promising for the fabrication of low resistance side contacts. More generally, controllable and stable doping of 2D semiconductors is a major challenge that must be overcome in the realization of both n-type and p-type semiconductor FETs. Enhancing our knowledge on doping and uniform growth of 2D semiconductors and heterostructures should enable the next generation of

low-power, high-speed devices, such as thin-tunnelling FETs, to be fabricated.

In addition to device-related challenges, uniform growth and environmental stability of 2D TMDs in their semiconducting and metallic phases are topics for future research. Although tremendous progress has been made in the uniform growth of graphene by chemical vapour deposition, large-area growth of two-element compounds (for example,  $\text{MoS}_2$ ) has proven more difficult. Although polycrystalline  $\text{MoS}_2$  and  $\text{WS}_2$  have been grown successfully using metal–organic chemical vapour deposition, the stoichiometry, chemical purity and yield must be optimized. A fact about TMDs that is frequently overlooked is that they are rarely stoichiometric or chemically pure as a consequence of impurities in the precursors. The ability to precisely control the phases and the number of layers of TMDs also remain challenges to be addressed. Epitaxial growth methods are currently being used to fabricate 2D materials for electronic applications. These methods could be particularly important for the development of 2D semiconductors for high performance FETs as well as thin-tunnel FETs that require high-quality heterostructures. In addition, large-area growth of high-quality 2D semiconductors by chemical vapour deposition or solution-phase exfoliation could fulfil the requirements for flexible electronics.

- Moore, G. E. Cramming more components onto integrated circuits. *Electronics* **38**, 114–177 (1965).
- Dennard, R. H., Gaensslen, F. H., Rideout, V. L., Bassous, E. & LeBlanc, A. R. Design of ion-implanted MOSFET's with very small physical dimensions. *IEEE J. Solid-State Circ.* **9**, 256–268 (1974).
- Mistry, K. *et al.* A 45 nm logic technology with high- $k^*$  metal gate transistors, strained silicon, 9 Cu interconnect layers, 193 nm dry patterning, and 100% Pb-free packaging. *IEEE Int. Electron Devices Meet.* 247–250 (IEEE, 2007).
- Cartwright, J. Intel enters the third dimension. *Nature* <http://www.nature.com/news/2011/110506/full/news.2011.274.html> (2011).
- Waldrop, M. M. The chips are down for Moore's law. *Nature* **530**, 144–147 (2016).
- Ferrain, I., Colinge, C. A. & Colinge, J.-P. Multi-gate transistors as the future of the classical metal-oxide-semiconductors field-effect-transistors. *Nature* **479**, 310–316 (2011).
- Colinge, J. P. Multiple-gate SOI MOSFETs. *Solid State Electron.* **48**, 897–905 (2004).
- The International Technology Roadmap for Semiconductors: 2012 Update*, <http://www.itrs2.net/> (ITRS, 2012).
- Del Alamo, J. A. Nanometer-scale electronics with III–V compound semiconductors. *Nature* **479**, 317–323 (2011).
- Colinge, J. P. in *FinFETs and Other Multi-Gate Transistors* (ed Colinge, J. P.) 1–48 (Springer, 2007).
- Huang, X. *et al.* Sub 50-nm FinFET: PMOS. *Tech. Dig. Int. Electron Devices Meet.* 67–70 (IEEE, 1999).
- Jan, C.-H. *et al.* A 22 nm SoC platform technology featuring 3-D tri-gate and high- $k^*$ /metal gate, optimized for ultra low power, high performance and high density SoC applications. *IEEE Int. Electron Devices Meet.* 3.1.1–3.1.4 (IEEE, 2012).
- Radosavljevic, M. *et al.* Electrostatics improvement in 3-D tri-gate over ultra-thin body planar InGaAs quantum well field effect transistors with high-K gate dielectric and scaled gate-to-drain/gate-to-source separation. *IEEE Int. Electron Devices Meet.* 33.1.1–33.1.4 (IEEE, 2011).
- Yu, B. *et al.* In Ultra-thin-body silicon-on-insulator MOSFETs for terabit-scale integration. *Proc. Int. Semiconductor Dev. Res. Symp.* 623–626 (Engineering Academic Outreach, 1997).
- Li, G.-W. *et al.* Ultrathin body GaN-on-insulator quantum well FETs with regrown ohmic contacts. *IEEE Electron Device Lett.* **33**, 661–663 (2012).
- Jena, D. Tunneling transistors based on graphene and 2D crystals. *Proc. IEEE* **101**, 1585–1602 (2013).
- Kang, J. H. *et al.* Graphene and beyond-graphene 2D crystals for next-generation green electronics. *Proc. SPIE* **9083**, 908305 (2014).
- Novoselov, K. S. *et al.* Electric field effect in atomically thin carbon films. *Science* **306**, 666–669 (2004).
- Avouris, P. *et al.* Graphene-based fast electronics and optoelectronics. *IEEE Int. Electron Devices Meet.* 23.1.1–23.1.4 (IEEE, 2010).
- Liao, L. *et al.* Sub-100 nm channel length graphene transistors. *Nano Lett.* **10**, 3952–3956 (2010).
- Liao, L. *et al.* High-speed graphene transistors with a self-aligned nanowire gate. *Nature* **467**, 305–308 (2010).
- Xia, F. *et al.* Ultrafast graphene photodetector. *Nat. Nanotechnol.* **4**, 839–843 (2009).
- Bonaccorso, F., Sun, Z., Hasan, T. & Ferrari, A. C. Graphene photonics and optoelectronics. *Nat. Photonics* **4**, 611–622 (2010).
- Dean, C. R. *et al.* Boron nitride substrates for high-quality graphene electronics. *Nat. Nanotechnol.* **5**, 722–726 (2010).
- Jena, D. & Konar, A. Enhancement in carrier mobility in semiconductor nanostructures by dielectric engineering. *Phys. Rev. Lett.* **98**, 136805 (2007).
- Cui, X. *et al.* Multi-terminal transport measurements of  $\text{MoS}_2$  using van der Waals heterostructure device platform. *Nat. Nanotechnol.* **10**, 534–540 (2015).
- Wang, Q. H., Kalantar-Zadeh, K., Kis, A., Coleman, J. N. & Strano, M. S. Electronics and optoelectronics of two-dimensional transition metal dichalcogenides. *Nat. Nanotechnol.* **7**, 699–712 (2012).
- Fiori, G. *et al.* Electronics based on two-dimensional materials. *Nat. Nanotechnol.* **9**, 768–779 (2014).
- Wang, H. *et al.* Integrated circuits based on bi-layer  $\text{MoS}_2$  transistors. *Nano Lett.* **12**, 4674–4680 (2012).
- Natori, K. Ballistic metal oxide semiconductor field effect transistor. *J. Appl. Phys.* **76**, 4879–4890 (1994).
- Natori, K. Scaling limit of the MOS transistor: a ballistic MOSFET. *JICICElect. Trans.* **E84C**, 1029–1036 (2001).
- Xu, X., Yao, W., Xiao, D. & Heinz, T. F. Spin and pseudospins in layered transition metal dichalcogenides. *Nat. Phys.* **10**, 343–350 (2014).
- Rodwell, M. *et al.* III–V FET channel designs for high current densities and thin inversion layers. *Device Res. Conf (DRC)* 149–152 (IEEE, 2010).
- Radosavljevic, B., Radenovic, A., Brivio, J., Giacometti, V. & Kis, A. Single-layer  $\text{MoS}_2$  transistors. *Nat. Nanotechnol.* **6**, 147–150 (2011).
- Fuhrer, M. S. & Hone, J. Measurement of mobility in dual-gate  $\text{MoS}_2$  transistor. *Nat. Nanotechnol.* **8**, 146–147 (2013).
- Wilson, J. A. & Yoffe, A. D. The transition metal dichalcogenide discussion and interpretation of the optical, electrical and structural properties. *Adv. Phys.* **18**, 193–335 (1969).
- Chhowalla, M. *et al.* The chemistry of two-dimensional layered transition metal dichalcogenide nanosheets. *Nat. Chem.* **5**, 263–275 (2013).
- Castro Neto, A. H., Guinea, F., Peres, N. M. R., Novoselov, K. S. & Geim, A. K. The electronic properties of graphene. *Rev. Mod. Phys.* **81**, 109 (2009).
- Schwierz, F. Graphene transistors: status, prospects, and problems. *Proc. IEEE* **101**, 1567–1584 (2013).
- Fang, T., Konar, A., Xing, H. & Jena, D. Mobility in semiconducting graphene nanoribbons: phonon, impurity, and edge roughness scattering. *Phys. Rev. B: Condens. Matter* **78**, 205403 (2008).
- Hwang, W. S. *et al.* Graphene nanoribbon field effect transistors on wafer scale epitaxial graphene on SiC substrates. *APL Mater.* **3**, 011101 (2015).
- Chau, R., Doyle, B., Datta, S., Kavalieros, J. & Zhang, K. Integrated nanoelectronics for the future. *Nat. Mater.* **6**, 810–812 (2007).
- Radosavljevic, M. *et al.* High-performance 40 nm gate length InSb p-channel compressively strained quantum well field effect transistors for low-power ( $V_{CC} = 0.5\text{V}$ ) logic applications. *IEEE Int. Electron Devices Meet.* 1–4 (IEEE, 2008).
- Podzorov, V., Gershenson, M. E., Kloc, Ch., Zeis, R. & Bucher, E. High-mobility field-effect transistors based on transition metal dichalcogenides. *Appl. Phys. Lett.* **84**, 3301 (2004).
- Mak, K. F., Lee, C., Hone, J., Shan, J. & Heinz, T. F. Atomically thin  $\text{MoS}_2$ : a new direct-gap semiconductor. *Phys. Rev. Lett.* **105**, 136805 (2010).
- Yoon, Y., Ganapathi, K. & Salahuddin, S. How good can monolayer  $\text{MoS}_2$  transistors be? *Nano Lett.* **11**, 3768 (2011).
- Fang, H. *et al.* High performance single layered  $\text{WSe}_2$  p-FETs with chemically doped contacts. *Nano Lett.* **12**, 3788–3792 (2012).

48. Wang, H. *et al.* Integrated circuits based on bilayer MoS<sub>2</sub> transistors. *Nano Lett.* **12**, 4674–4680 (2012).
49. Das, S., Chen, H.-Y., Penumatcha, A. V. & Appenzeller, J. High performance multilayer MoS<sub>2</sub> transistors with scandium contacts. *Nano Lett.* **13**, 100–105 (2012).
50. Larentis, S., Fallahazad, B. & Tutuc, E. Field-effect transistors and intrinsic mobility in ultra-thin MoSe<sub>2</sub> layers. *Appl. Phys. Lett.* **101**, 223104 (2012).
51. Radisavljevic, B. & Kis, A. Mobility engineering and a metal–insulator transition in monolayer MoS<sub>2</sub>. *Nat. Mater.* **12**, 815–820 (2013).
52. Fang, H. *et al.* Degenerate n-doping of few layered transition metal dichalcogenides by potassium. *Nano Lett.* **13**, 1991–1995 (2013).
53. Du, Y. *et al.* MoS<sub>2</sub> field-effect transistors with graphene/metal heterocontacts. *IEEE Electron Device Lett.* **35**, 599–601 (2014).
54. Allain, A. & Kis, A. Electron and hole mobilities in single layer WSe<sub>2</sub>. *ACS Nano* **8**, 7180–7185 (2014).
55. Jo, S., Ubrig, N., Berger, H., Kuzmenko, A. B. & Morpurgo, A. F. Mono- and bilayer WS<sub>2</sub> light-emitting transistors. *Nano Lett.* **14**, 2019–2025 (2014).
56. Lin, Y.-F. *et al.* Ambipolar MoTe<sub>2</sub> transistors and their applications in logic circuits. *Adv. Mater.* **26**, 3263–3269 (2014).
57. Pradhan, N. R. *et al.* Ambipolar molybdenum diselenide field-effect transistors: field effect and Hall mobilities. *ACS Nano* **8**, 7923–7929 (2014).
58. Jariwala, D., Sangwan, V. K., Lauhon, L. J., Marks, T. J. & Hersam, M. C. Emerging device applications for semiconducting two-dimensional transition metal dichalcogenides. *ACS Nano* **8**, 1102–1120 (2014).
59. Gong, K. *et al.* Electric control of spin in monolayer WSe<sub>2</sub> field effect transistors. *Nanotechnology* **25**, 435201 (2014).
60. Kappera, R. *et al.* Phase-engineered low-resistance contacts for ultrathin MoS<sub>2</sub> transistors. *Nat. Mater.* **13**, 1128–1134 (2014).
61. Schimdt, H., Giustiniano, F. & Eda, G. Electronic transport properties of transition metal dichalcogenide field-effect devices: surface and interface effects. *Chem. Soc. Rev.* **44**, 7715–7736 (2015).
62. Liu, H., Neal, A. T., Zhu, Z., Tomanek, D. & Ye, P. D. Phosphorene: an unexplored 2D semiconductor with a high hole mobility. *ACS Nano* **8**, 4033–4041 (2014).
63. Koenig, S. P. *et al.* Electric field effect in ultrathin black phosphorus. *Appl. Phys. Lett.* **104**, 103106 (2014).
64. Li, L. *et al.* Black phosphorus field-effect transistors. *Nat. Nanotechnol.* **9**, 372–377 (2014).
65. Buscema, M. *et al.* Fast and broadband photoresponse of few-layer black phosphorus field-effect transistors. *Nano Lett.* **14**, 3347–3352 (2014).
66. Xia, F., Wang, H. & Jia, Y. Rediscovering black phosphorus: a unique anisotropic 2D material for optoelectronics and electronics. *Nat. Commun.* **5**, 4458 (2014).
67. Das, S. *et al.* Tunable transport gap in phosphorene. *Nano Lett.* **14**, 5733–5739 (2014).
68. Liu, H. *et al.* The effect of dielectric capping on few-layer phosphorene transistors: tuning the Schottky barrier heights. *IEEE Electron Device Lett.* **35**, 795–797 (2014).
69. Deng, Y. *et al.* Black phosphorus–monolayer MoS<sub>2</sub> van der Waals heterojunction p–n diode. *ACS Nano* **8**, 8292–8299 (2014).
70. Wang, H. *et al.* Black phosphorus radio-frequency transistors. *Nano Lett.* **14**, 6424–6429 (2014).
71. Haratipour, N., Robbins, M. C. & Koester, S. J. Black phosphorus p-MOSFETs with 7-nm HfO<sub>2</sub> gate dielectric and low contact resistance. *IEEE Electron. Device Lett.* **36**, 411–413 (2015).
72. Du, Y. *et al.* Device perspective for black phosphorus field-effect transistors: contact resistance, ambipolar behavior, and scaling. *ACS Nano* **8**, 10035–10042 (2014).
73. Xiong, K., Luo, X. & Huang, J. C. M. Phosphorene FETs — Promising transistors based on a few layers of phosphorus atoms. *IEEE MTT-S Int. Microwave Workshop Ser. Adv. Mater. Processes RF THz Appl.* 1–3 (IEEE, 2015).
74. Tao, L. *et al.* Silicene field effect transistors operating at room temperature. *Nat. Nanotechnol.* **10**, 227–231 (2015).
75. Splendiani, A. *et al.* Emerging photoluminescence in monolayer MoS<sub>2</sub>. *Nano Lett.* **10**, 1271–1275 (2010).
76. Jariwala, B. *et al.* Synthesis and characterization of ReS<sub>2</sub> and ReSe<sub>2</sub> layered chalcogenide single crystals. *Chem. Mater.* **28**, 3352–3359 (2016).
77. Frindt, R. F. Superconductivity in ultrathin NbSe<sub>2</sub> layers. *Phys. Rev. Lett.* **28**, 299–301 (1971).
78. Ye, J. T. *et al.* Superconducting dome in a gate-tuned band insulator. *Science* **338**, 1193–1196 (2012).
79. Sipos, B. *et al.* From Mott state to superconductivity in 1T-TaS<sub>2</sub>. *Nat. Mater.* **7**, 960–965 (2008).
80. Ayari, A., Cobas, E., Ogundadege, O. & Fuhrer, M. S. Realization and electrical characterization of ultrathin crystals of layered transition-metal dichalcogenides. *J. Appl. Phys.* **101**, 014507 (2007).
81. Liu, L., Lu, Y. & Guo, J. On monolayer MoS<sub>2</sub> field-effect transistors at the scaling limit. *IEEE Trans. Electron Devices* **60**, 4133–4139 (2013).
82. Alam, K. & Lake, R. Monolayer MoS<sub>2</sub> transistors beyond the technology road map. *IEEE Trans. Electron Devices* **59**, 3250–3254 (2012).
83. Kim, S. *et al.* High-mobility and low-power thin-film transistors based on multilayer MoS<sub>2</sub> crystals. *Nat. Commun.* **3**, 1011 (2012).
84. Enyashin, A. N. & Seifert, G. Electronic properties of MoS<sub>2</sub> monolayer and related structures. *Nanosyst. Phys. Chem. Math.* **5**, 517–539 (2014).
85. McDonnell, S. *et al.* Defect dominated doping and contact resistance in MoS<sub>2</sub>. *ACS Nano* **8**, 2880–2888 (2014).
86. Voiry, D. *et al.* The role of electronic coupling between substrate and 2D MoS<sub>2</sub> nanosheets in electrocatalytic production of hydrogen. *Nat. Mater.* <http://dx.doi.org/10.1038/nmat4660> (2016).
87. Hwang, W. S. *et al.* Transistors with chemically synthesized layered semiconductor WS<sub>2</sub> exhibiting 10<sup>5</sup> room temperature modulation and ambipolar behavior. *Appl. Phys. Lett.* **101**, 013107 (2012).
88. Jena, D., Banerjee, K. & Xing, G. H. 2D crystal semiconductors: Intimate contacts. *Nat. Mater.* **13**, 1076–1078 (2014).
89. Allain, A., Kang, J., Kis, A. & Banerjee, K. Electrical contacts in two-dimensional semiconductors. *Nat. Mater.* **14**, 1195–1205 (2015).
90. Yoon, J. *et al.* Highly flexible and transparent multilayer MoS<sub>2</sub> transistors with graphene electrodes. *Small* **9**, 3295–3300 (2013).
91. Das, S., Gulott, R., Sumant, A. V. & Roelofs, A. All two-dimensional, flexible, transparent, and thinnest thin film transistor. *Nano Lett.* **14**, 2861–2866 (2014).
92. Eda, G. *et al.* Coherent atomic and electronic heterostructures of single layer MoS<sub>2</sub>. *ACS Nano* **6**, 7311–7317 (2012).
93. Cho, S. *et al.* Phase patterning of ohmic homojunction in MoTe<sub>2</sub>. *Science* **348**, 625–628 (2015).
94. Akinwande, D., Petrone, N. & Hone, J. Two-dimensional flexible nanoelectronics. *Nat. Commun.* **5**, 5678 (2014).
95. Pu, J. *et al.* Highly flexible MoS<sub>2</sub> thin-film transistors with ion gel dielectrics. *Nano Lett.* **12**, 4013–4017 (2012).
96. Chang, H.-Y. *et al.* High-performance, highly bendable MoS<sub>2</sub> transistors with high-K dielectrics for flexible low-power systems. *ACS Nano* **7**, 5446–5452 (2013).
97. Zhu, W. *et al.* Flexible black phosphorus ambipolar transistors, circuits and AM demodulator. *Nano Lett.* **15**, 1885–1890 (2015).
98. Roy, T. *et al.* Field-effect transistors built from all two-dimensional material components. *ACS Nano* **8**, 6259–6264 (2014).
99. Kang, K. *et al.* High-mobility three-atom-thick semiconducting films with wafer-scale homogeneity. *Nature* **520**, 656–660 (2015).
100. Bridgman, P. M. Two new modifications of phosphorus. *J. Am. Chem. Soc.* **36**, 1344–1363 (1914).
101. Hultgren, R., Gingrich, N. S. & Warren, B. E. The atomic distribution in red and black phosphorus and the crystal structure of black phosphorus. *J. Chem. Phys.* **3**, 351–355 (1935).
102. Keys, R. W. The electrical properties of black phosphorus. *Phys. Rev.* **92**, 580–584 (1953).
103. Takao, Y., Asahina, H. & Morita, A. Electronic structure of black phosphorus in tight binding approach. *J. Phys. Soc. Jpn.* **50**, 3362–3369 (1981).
104. Akahama, Y., Endo, S. & Narita, S. Electrical properties of black phosphorus single crystals. *J. Phys. Soc. Jpn.* **52**, 2148–2155 (1983).
105. Youngblood, N., Chen, C., Koester, S. J. & Li, M. Waveguide integrated black phosphorus photodetector with high responsivity and low dark current. *Nat. Photonics* **9**, 247–252 (2015).
106. Castellanos-Gomez, A. *et al.* Isolation and characterization of few-layer black phosphorus. *2D Mater.* **1**, 025001 (2014).
107. Gillgren, N. *et al.* Gate tunable quantum oscillations in air-stable and high mobility few-layer phosphorene heterostructures. *2D Mater.* **2**, 011001 (2015).
108. Li, L. *et al.* Quantum oscillations in two-dimensional electron gas in black phosphorus thin films. *Nat. Nanotechnol.* **10**, 608–613 (2015).
109. Island, J. O. *et al.* Environmental stability of few-layer black phosphorus. *2D Mater.* **2**, 011002 (2015).
110. Kim, J. S. *et al.* Toward air-stable multilayer phosphorene thin-films and transistors. *Sci. Rep.* **5**, 8989 (2015).
111. Du, Y. *et al.* *Ab initio* studies on atomic and electronic structures of black phosphorus. *J. Appl. Phys.* **107**, 093718 (2010).
112. Avsar, A. *et al.* Air-stable transport in graphene-contacted, fully encapsulated ultrathin black phosphorus-based field-effect transistors. *ACS Nano* **9**, 4138–4145 (2015).
113. Na, J. *et al.* Few-layer black phosphorus field-effect transistors with reduced current fluctuation. *ACS Nano* **8**, 11755–11762 (2014).
114. Wood, J. D. *et al.* Effective passivation of exfoliated black phosphorus transistors against ambient degradation. *Nano Lett.* **14**, 6964–6970 (2014).
115. Takeda, K. & Shiraiishi, K. Theoretical possibility of stage corrugation in Si and Ge analogs of graphite. *Phys. Rev. B: Condens. Matter* **50**, 14916–14922 (1994).
116. Cinquanta, E. *et al.* Getting through the nature of silicene: an sp<sup>2</sup>–sp<sup>3</sup> two-dimensional silicon nanosheet. *J. Phys. Chem. C* **117**, 16719–16724 (2013).
117. Vogt, P. *et al.* Silicene: compelling experimental evidence for graphene-like two-dimensional silicon. *Phys. Rev. Lett.* **108**, 155501 (2012).
118. Li, X. *et al.* Intrinsic electrical transport properties of monolayer silicene and MoS<sub>2</sub> from first principles. *Phys. Rev. B: Condens. Matter* **87**, 115418 (2013).
119. Nikonov, D. E. & Young, I. A. Benchmarking of beyond CMOS exploratory devices for logic integrated circuits. *IEEE J. Explor. Solid-State Comput. Devices Circuits* **1**, 3–11 (2015).
120. Li, M. O., Esseni, D., Nahas, J. J., Jena, D. & Xing, H. G. Two-dimensional heterojunction interlayer tunneling field effect transistors (Thin-TFETs). *IEEE J. Electron Devices Soc.* **3**, 200–207 (2015).
121. Theis, T. N. & Solomon, P. N. In quest of the next switch: prospects for greatly reduced power dissipation in a successor to the silicon field effect transistor. *Proc. IEEE* **98**, 2005–2014 (2010).
122. Gnani, E., Maiorano, P., Reggiani, S., Gnudi, A. & Baccarani, G. Investigation on superlattice heterostructures for steep-slope nanowire FETs. *Device Res. Conf. (DRC)* 201–202 (IEEE, 2011).
123. Zhang, Q., Zhao, W. & Seabaugh, A. Low-threshold swing-tunnel transistors. *IEEE Electron Device Lett.* **27**, 297–300 (2006).
124. Lu, H. & Seabaugh, A. Tunnel field-effect transistors: state-of-the-art. *IEEE J. Electron Devices Soc.* **2**, 44–49 (2014).
125. Zhou, G. *et al.* Novel gate-recessed vertical InAs/GaSb TFETs with record high ION of 180 μA/μm at VDS = 0.5 V. *IEEE Int. Electron Devices Meet.* 32.6.1–32.6.4 (IEEE, 2012).
126. Lin, Y. C. *et al.* Atomically thin resonant tunnel diodes built from synthetic van der Waals heterostructures. *Nat. Commun.* **6**, 7311 (2014).
127. Yan, R. *et al.* Esaki diodes in van der Waals heterojunctions with broken gap energy band alignment. *Nano Lett.* **15**, 5791–5798 (2015).
128. Roy, T. *et al.* Dual-gated MoS<sub>2</sub>/WSe<sub>2</sub> van der Waals tunnel diodes and transistors. *ACS Nano* **9**, 2071–2079 (2015).
129. Sarkar, D. *et al.* A sub-thermionic tunnel field effect transistor with an atomically thin channel. *Nature* **526**, 91–95 (2015).

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#### Competing interests

The authors declare no competing interests.

#### SUPPLEMENTARY INFORMATION

See online article: S1 (box) | S2 (figure)

**Box 1. Quantitative description of FET**

Here we quantitatively describe the fundamental operating principles of FETs to highlight the advantages of 2D semiconductors. A typical FET with a single-layer TMD semiconductor channel and its operating mechanisms are schematically described in FIG. 2 of the manuscript. For modal distribution of electrons in the 2D  $(k_x, k_y)$ -space such as those shown in Figure 2d, the net carrier density is

$$n_s = \frac{g_s \cdot g_v}{(2\pi)^2} \frac{m_c^* kT}{\hbar^2} \pi \ln[(1 + \eta_s)(1 + \eta_d)], \tag{1}$$

where  $g_s$  and  $g_v$  are the spin and valley degeneracy values,  $m_c^*$  is the effective mass of carriers,  $(E_{Fs} - E_c)/kT = \eta_s$  and  $(E_{Fd} - E_c)/kT = \eta_d$  ( $E_{Fs}$  and  $E_{Fd}$  is the quasi-Fermi-level of the source and drain electrodes,  $E_c$  is the energy of the conduction band) are the important dimensionless source and drain degeneracy terms. When  $V_{ds} = 0$ ,  $E_{Fs} = E_{Fd}$ , the relation becomes

$$n_s = D_0 kT \ln[1 + \exp[\eta_s]], \tag{2}$$

where the 2D band-edge density of states is

$$D_0 = g_s \cdot g_v \frac{m_c^*}{2\pi\hbar^2} \tag{3}$$

We identify the degeneracy energy as  $\eta_s = \ln[e^{n_s/n_q} - 1]$ , where  $n_q = D_0 kT = C_q V_{th} / q$  is a characteristic 2D carrier concentration, defined by the product of the quantum capacitance  $C_q = q^2 D_0$  and the thermal voltage. A characteristic carrier density from the electrostatic capacitance is  $n_b = C_b V_{th} / q$ . With these connections, we rewrite the relation of the carrier density with the gate voltage as

$$e^{n_s/n_b} (e^{n_q/n_b} - 1) = e^{\frac{V_{gs} - V_T}{V_{th}}} \tag{4}$$

the solution to which gives the mobile carrier density  $n_s(V_{gs})$  in the semiconductor channel as a function of the gate voltage  $V_{gs}$ .

The drain current

$$I_d / W = J_0 [F_{1/2}(\eta_s) - F_{1/2}(\eta_d)] \quad (5)$$

is obtained by summing the contribution from all modes. Here the coefficient for the current density per unit width of the transistor is  $J_0 = qn_q v_{inj}$ , with  $n_q$  the 2D quantum concentration,

$v_{inj} = \sqrt{2kT / \pi^2 m_c^*}$  a characteristic velocity, and  $F_j(\eta) = \int_0^\infty \frac{u^j}{1 + e^{u-\eta}}$  are Fermi-Dirac integrals of

order  $j$  that depend on the terminal voltages  $\eta_s = (E_{Fs} - E_c) / kT$  and  $\eta_d = (E_{Fd} - E_c) / kT$ , with  $\eta_s - \eta_d = qV_{ds} / kT = v_d$ . In an electrostatically well-designed FET, at the source-injection point, the gate is in complete control over the 2D carrier density. So we also have

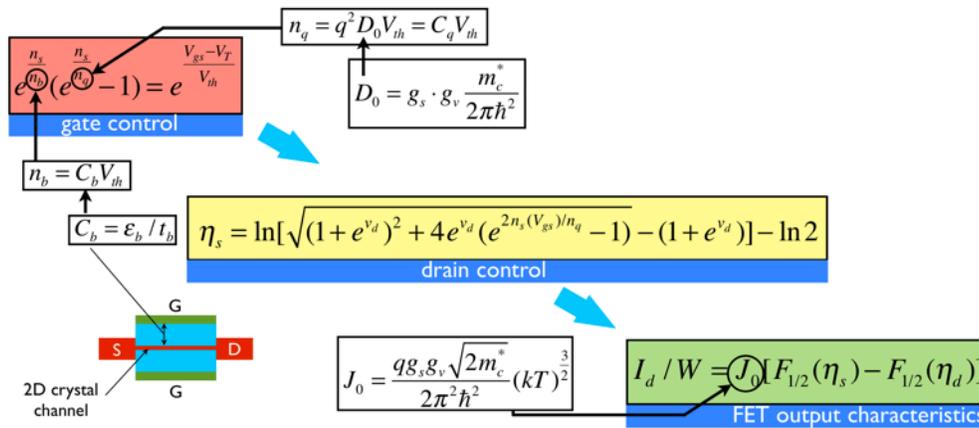
$$n_s = \frac{D_0}{2} kT \ln[(1 + \exp[\eta_s])(1 + \exp[\eta_s - v_d])] \quad (6)$$

This leads to a self-consistent solution

$$\eta_s = \ln[\sqrt{(1 + e^{v_d})^2 + 4e^{v_d}(e^{2n_s(V_{gs})/n_q} - 1)} - (1 + e^{v_d})] - \ln 2 \quad (7)$$

where  $n_s(V_{gs})$  is the solution to the central equation (4).

The three equations capture the entire output characteristics of the ballistic FET. Neither the channel length  $L$ , nor the carrier mobility appear in the ballistic FET current because there is no scattering. The process for mathematical evaluation of each FET component to extract the FET performance is summarized in Figure S1. The solutions can be evaluated easily on a computer and discussed in the main manuscript.



**Figure 1. Flowchart for assessing performance of FETs**

Flowchart for evaluating the key parameters of a ballistic FET consisting of a 2D channel, shown schematically in the lower left corner (G = gate, S = source, D = drain, blue region is the gate dielectric). The application of a gate bias  $V_{gs}$  introduces a gate capacitance ( $C_b$ ), which in turn results characteristic carrier density in the channel ( $n_b$ ) above the threshold voltage. The tuning of the carrier concentration via modulation of gate voltage results in an expression that relates the mobile carrier density  $n_s(V_{gs})$  in the semiconductor channel as a function of the gate voltage  $V_{gs}$  [equation (4)], which includes the 2D carrier concentration ( $n_q$ ) term that is related to the density of states at the band edge of the 2D semiconductor ( $D_0$ ). The influence of the drain voltage,  $V_{ds}$ , on the can be obtained from the dimensionless degeneracy source term ( $\eta_s$ ). From the gate and drain electrostatics, it is possible to extract the FET operation characteristics in terms of the drain current with gate voltage from the equation shown in the bottom right corner.