

# 246 nm AlN-delta-GaN Quantum Well Ultraviolet Light-Emitting Diode

Cheng Liu<sup>1‡</sup>, Yu Kee Ooi<sup>1</sup>, SM Islam<sup>2</sup>, Huili (Grace) Xing<sup>2,3</sup>, Debdeep Jena<sup>2,3</sup> and Jing Zhang<sup>1‡</sup>

<sup>1</sup>Department of Electrical and Microelectronics Engineering, Rochester Institute of Technology, Rochester, NY 14623, USA

<sup>2</sup>School of Electrical and Computer Engineering, Cornell University, Ithaca, NY 14853, USA

<sup>3</sup>Department of Material Science and Engineering, Cornell University, Ithaca, NY 14853, USA

<sup>‡</sup>Email: [cl7007@rit.edu](mailto:cl7007@rit.edu), <sup>‡</sup>Email: [Jing.Zhang@rit.edu](mailto:Jing.Zhang@rit.edu)

**Abstract:** The 246 nm AlN-delta-GaN quantum well ultraviolet light-emitting diode was proposed and realized experimentally, with the dominant transverse electric-polarized emission been verified by both the  $k\cdot p$  simulation and the room-temperature polarization-dependent electroluminescence measurements.

**OCIS codes:** (230.3670) Light-emitting diodes; (230.5590) Quantum-well, -wire and -dot devices

High-efficiency deep ultraviolet (UV) light emitting diodes (LEDs) operating at wavelengths ( $\lambda$ ) below 300 nm are required for a wide variety of important applications such as sterilization, medical diagnostics and water purification. High Al-content AlGaIn quantum wells (QWs) are employed to pursue such deep-UV LEDs and have been widely studied and reported for 220-280 nm spectral regimes [1-3]. Nevertheless, UV LEDs with AlGaIn active regions usually suffer from low external quantum efficiencies (<10%), especially at shorter emission wavelengths [3]. The efficiency in such deep-UV LEDs is limited by the poor carrier injection efficiency caused by the low electrical conductivity in p- and n-types AlGaIn layers; low internal quantum efficiency ( $\eta_{IQE}$ ) resulted from reduced optical matrix element; and the low light extraction efficiency. Furthermore, the valence subband crossover in conventional AlGaIn QWs and their role in determining the direction of optical emission is of crucial importance [2, 4]. Thus, it is important to explore novel QW active regions which can suppress the challenges and lead to improved quantum efficiencies.

Our previous work has proposed the use of the AlN-delta-GaN QW structure with 3-4 monolayers of the delta-GaN layer for 298 nm UV LEDs to reduce degradation due to the Quantum Confined Stark Effect, when compared to conventional AlGaIn QWs that emit at a similar wavelength [5]. Pushing to shorter wavelengths such as  $\sim$ 250 nm, the use of conventional AlGaIn QWs suffer from significant band-mixing effect as the Al-content is near the crossover point beyond which the energy separation between heavy hole (HH) and crystal-field split-off hole (CH) leads to both low transverse electric (TE)- and transverse magnetic (TM)-polarized optical emission [4]. Thus, in this work, we propose the use of the AlN-delta-GaN QW with thinner delta-GaN thickness in order to address the band-mixing effect with  $\lambda\sim$ 250 nm. The valence band structure and corresponding polarization-dependent spontaneous emission of the proposed AlN-delta-GaN QW was investigated by the 6-band  $k\cdot p$  simulations [4, 8-9]. The AlN-delta-GaN QW-like UV LEDs with 2-3 monolayers of delta-GaN layer were grown on the AlN/sapphire template by plasma-assisted molecular beam epitaxy (MBE) [6-7]. The angle-dependent and polarization-dependent electroluminescence (EL) measurements were conducted to validate the physics from the proposed QW design.

A schematic layer structure of the AlN-delta-GaN QW is shown in figure 1 which includes the 2-nm AlN sub-QW region with 4-Å delta-GaN layer to aim for  $\sim$ 250 nm peak emission. Here, the AlN sub-QW and the AlN barrier regions are clearly separated into the  $k\cdot p$  simulations. From the calculated valence band structure, we identify two major advantages of using the AlN-delta-GaN QW: 1) the HH subband has been flipped into the topmost position, and 2) there exists a large separation between the HH1 and CH1 subbands which successfully resolves the band mixing issue. Thus, it is expected that the dominant conduction band (C) - HH1 transition will be endured by the AlN-delta-GaN QW active region. As a result, from the spontaneous emission spectra in figure 1, dominant TE-polarized spontaneous emission is obtained with  $\lambda = 243$  nm.

To verify the polarization properties and the physics of the proposed AlN-delta-GaN QWs, the MBE-grown UV LED with such active region was investigated in this study. As shown in figure 2, the active region consists of 2-3 monolayers delta-GaN layer embedded in 1.75 nm AlN layer with graded AlGaIn p- and n-polarization-induced-doped cladding layers. More MBE growth and device fabrication details can be found in *Refs.* 6-7. Room temperature EL measurement was first carried out, showing that the peak emission wavelength is 246 nm with current density ( $j$ ) of 16.67 A/cm<sup>2</sup> from the AlN-delta-GaN QW with 2-3 monolayers delta-GaN layer.

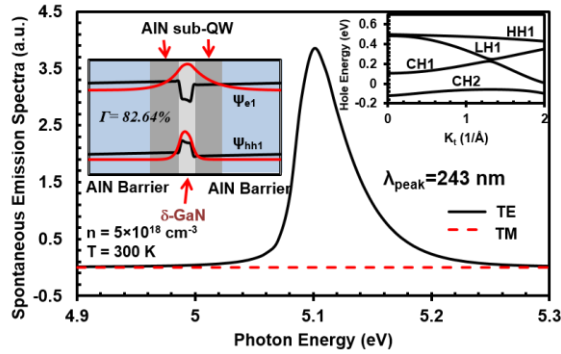


Figure 1: TE- and TM-polarized spontaneous emission spectra at  $n = 5 \times 10^{18} \text{ cm}^{-3}$ . Inset left: band structure of AlN-delta-GaN QW structure with electron and hole wavefunctions; inset right: valence band structure.

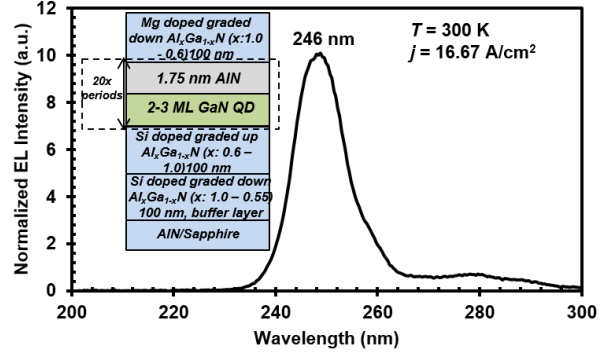


Figure 2: EL spectrum of MBE-grown AlN-delta-GaN QW-like LED with peak emission wavelength of  $246 \text{ nm}$  at  $j = 16.67 \text{ A/cm}^2$ . Inset: schematic of MBE-grown AlN-delta-GaN QW LED.

The polarization-dependent EL measurements were performed with the setup plotted in figure 3(a), similar with Refs. 2, 5, 10. A Glan-Taylor polarizer is placed between the stationary stage and the optical fiber and can be rotated to resolve the  $I_{\parallel}$  (electric field parallel to the incident plane) and  $I_{\perp}$  (electric field perpendicular to the incident plane). At a particular detection angle ( $\theta$ ), TE and TM components can be derived by the equations  $I_{\parallel} = I_{\text{TEy}} \cos^2 \theta + I_{\text{TM}} \sin^2 \theta$  and  $I_{\perp} = I_{\text{TEy}}$ . Figure 3(b) presents the polarization-dependent EL spectra at  $\theta = 30^\circ$  for the AlN-delta-GaN QW LED. The results show that the  $I_{\perp}/I_{\parallel}$  ratio is 1.18 for  $\theta = 30^\circ$  with the corresponding TE/TM ratio calculated as 17.731. Thus, dominant TE-polarized emission is demonstrated from the AlN-delta-GaN QW structure which can be attributed to the dominant C-HH transition from the proposed design.

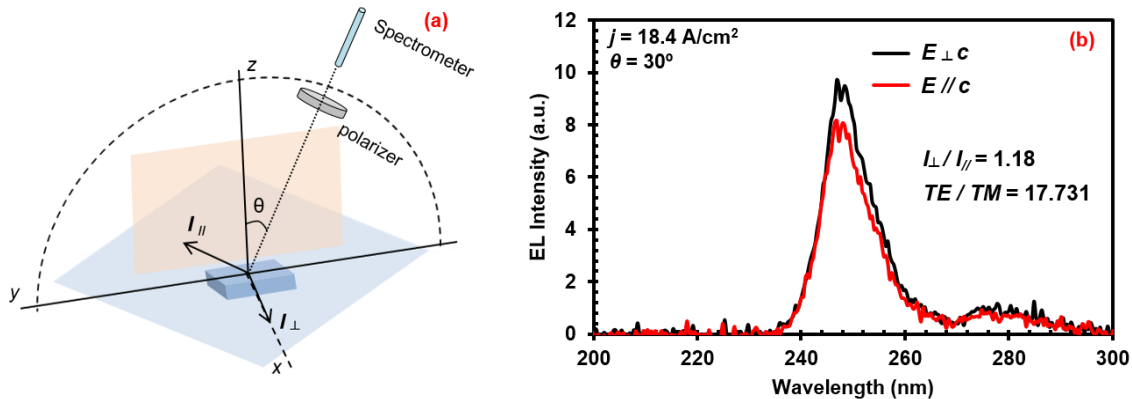


Figure 3: (a) Schematic of polarization-dependent EL measurement setup; (b) Polarization-dependent EL measurement at  $\theta = 30^\circ$ .

In summary, the AlN-delta-GaN QW is proposed, and experimentally demonstrated for  $246 \text{ nm}$  UV LED, and the polarization properties are investigated for the first time. The use of the AlN-delta-GaN QW with 2-3 monolayers delta-GaN layer addresses the band mixing problem, and leads to dominant C-HH transition. As a result, dominant TE-polarized emission is demonstrated by polarization-dependent EL measurements. Thus, it is expected that the use of the AlN-delta-GaN QW will serve as a promising alternative for  $250 \text{ nm}$  UV LEDs.

## References

- [1] H. Hirayama, N. Maeda, S. Fujikawa, S. Toyoda, and N. Kamata, Jpn. J. Appl. Phys., 53, 100209 (2014).
- [2] T. Kolbe, A. Knauer, C. Chua, Z. Yang, H. Rodrigues, S. Einfeldt, P. Vogt, N. M. Johnson, M. Weyers and M. Kneissl, Appl. Phys. Lett., 97, 171105 (2010).
- [3] F. Mehnke, C. Kuhn, M. Guttmann, C. Reich, T. Kolbe, V. Kueller, A. Knauer, M. Lapeyrade, S. Einfeldt and, J. Rass, T. Wernicke, M. Weyers, and M. Kneissl, Appl. Phys. Lett., 105, 051113 (2014).
- [4] J. Zhang, H. Zhao, and N. Tansu, Appl. Phys. Lett., 97, 111105 (2010).
- [5] C. Liu, Y. K. Ooi, S. M. Islam, J. Verma, H. Xing, D. Jena and J. Zhang, Appl. Phys. Lett., (submitted)
- [6] S. M. Islam, V. Protasenko, S. Rouvimov, H. Xing, and D. Jena, Jpn. J. Appl. Phys. 55, 05FF06 (2016).
- [7] J. Verma, S. M. Islam, V. Protasenko, P. K. Kandaswamy, H. Xing, and D. Jena, Appl. Phys. Lett., 104, 021105 (2014).
- [8] C. Liu, Y. K. Ooi, and J. Zhang, J. Appl. Phys. 119, 083102 (2016).
- [9] J. Zhang, H. Zhao, and N. Tansu, Appl. Phys. Lett. 98, 171111 (2011).
- [10] K. H. Li, Q. Wang, H. P. T. Nguyen, S. Zhao and Z. Mi, Phys. Status Solidi, 212, No. 5, 941 (2015)