

Two-carrier model-fitting of Hall effect in semiconductors with dual-band occupation: A case study in GaN two-dimensional hole gas

Cite as: J. Appl. Phys. 137, 025702 (2025); doi: 10.1063/5.0248998
Submitted: 14 November 2024 · Accepted: 20 December 2024 ·
Published Online: 9 January 2025



Joseph E. Dill,^{1,a)} Chuan F. C. Chang,² Debdeep Jena,^{3,4,5} and Huili Grace Xing^{3,4,5,a)}

AFFILIATIONS

¹School of Applied and Engineering Physics, Cornell University, Ithaca, New York 14853, USA

²Department of Physics, Cornell University, Ithaca, New York 14853, USA

³School of Electrical and Computer Engineering, Cornell University, Ithaca, New York 14853, USA

⁴Department of Materials Science and Engineering, Cornell University, Ithaca, New York 14853, USA

⁵Kavli Institute at Cornell for Nanoscale Science, Cornell University, Ithaca, New York 14853, USA

^{a)}Authors to whom correspondence should be addressed: jed296@cornell.edu and grace.xing@cornell.edu

ABSTRACT

We develop a two-carrier Hall effect model-fitting algorithm to analyze temperature-dependent magnetotransport measurements of a high-density ($\sim 4 \times 10^{13} \text{ cm}^{-2}$) polarization-induced two-dimensional hole gas (2DHG) in a GaN/AlN heterostructure. Previous transport studies in GaN 2DHGs have reported a twofold reduction in 2DHG carrier density when cooled from room to cryogenic temperature. We demonstrate that this apparent drop in carrier density is an artifact of assuming one species of charge carrier when interpreting Hall effect measurements. Using an appropriate two-carrier model, we resolve light hole (LH) and heavy hole (HH) carrier densities congruent with self-consistent Poisson-k-p simulations and observe an LH mobility of $\sim 1400 \text{ cm}^2/\text{Vs}$ and HH mobility of $\sim 300 \text{ cm}^2/\text{Vs}$ at 2 K. This report constitutes the first experimental signature of LH band conductivity reported in GaN.

© 2025 Author(s). All article content, except where otherwise noted, is licensed under a Creative Commons Attribution (CC BY) license (<https://creativecommons.org/licenses/by/4.0/>). <https://doi.org/10.1063/5.0248998>

I. INTRODUCTION

P-type polarization doping in metal-polar GaN/Al(GaN) heterostructures has been developed¹ as an alternative to chemical doping, which suffers from the lack of a shallow acceptor dopant in GaN (activation energy 135–170 meV for Mg^{2–5}). Polarization discontinuity at the hetero-interface, combined with piezoelectric charge from epitaxial strain of the GaN layer, can produce a two-dimensional hole gas (2DHG) confined to the GaN,⁶ even without intentional chemical dopants, as recently demonstrated by Chaudhuri *et al.*⁵ and Beckmann *et al.*⁷

In a singular undoped GaN/AlN heterojunction,^{5,8} the 2DHG density is $\sim 4 \times 10^{13} \text{ cm}^{-2}$. The lack of parasitic electron channels and ionized impurities from chemical dopants in this material stack greatly minimizes extrinsic carrier scattering, boosting cryogenic hole mobility. Recent growths of GaN/AlN 2DHGs by molecular beam epitaxy (MBE) have also utilized

buried impurity blocking layers⁸ and single-crystal AlN substrates⁹ to reduce impurity and dislocation scattering, respectively, enabling a sheet resistance in a GaN 2DHG below $1 \text{ k}\Omega/\square$, and reported mobility above $280 \text{ cm}^2/\text{Vs}$ at 10 K.⁹ This high mobility (relative to what has previously been reported in p-type GaN) makes this material system well-suited to study hole transport phenomenon in GaN. However, the high carrier density in the GaN/AlN 2DHG presents experimental complications to accurately characterizing its transport.

Classical Hall effect measurements¹⁰ are a standard means of characterizing transport in semiconductors. With a source current I , the sheet resistance is determined from a zero magnetic field measurement of the longitudinal voltage V_x ,

$$R_{sh} = fR_{xx} = f \frac{V_x}{I} = \frac{1}{qn\mu}, \quad (1)$$

09 January 2025 13:58:58

where q is the electron charge, f is a geometric factor, and the transverse voltage V_y under a perpendicular magnetic field B gives the Hall coefficient,

$$R_H = \frac{R_{xy}}{B} = \frac{V_y}{IB} = \frac{1}{qn}. \quad (2)$$

These equations decouple the free carrier density $n = 1/qR_H$ and mobility $\mu = R_H/R_{sh}$ in terms of the measured quantities. Because V_y is expected to vary linearly with B [see Eq. (2)], most quick-feedback Hall effect measurement systems only measure V_y at one or two low magnetic field points to calculate R_{xy} (typically ~ 0.5 T for a tabletop Hall setup).

However, these equations assume a single-carrier population with uniform mobility. Due to its high carrier density, self-consistent Poisson-k-p simulations of the GaN/AlN 2DHG place the Fermi level 40–60 meV below the heavy hole (HH) Γ -point and 25–35 meV below the light hole (LH) Γ -point, degenerately occupying both bands¹¹ (see band structure inset in Fig. 1). The LH

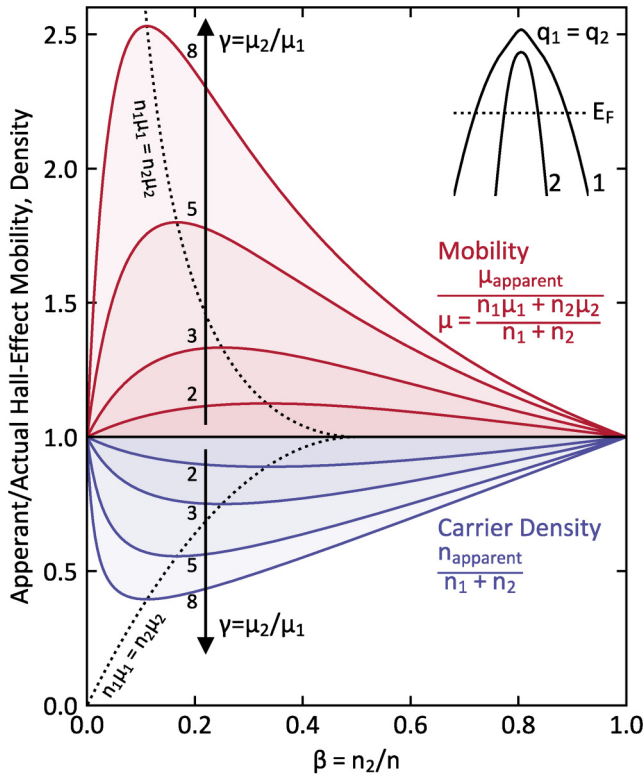


FIG. 1. Ratio between the apparent Hall mobility and true mobility $\mu_{\text{apparent}}/\mu$ (red), and likewise for the carrier density n_{apparent}/n (blue), obtained from a single-carrier interpretation of low-field Hall effect measurements of a system of two conducting channels with the same charge polarity ($q_1 = q_2$): $n = n_1 + n_2$; $\mu = (n_1\mu_1 + n_2\mu_2)/n$. Both are plotted as a function of the fraction of carrier in the 2nd channel $\beta = n_2/n$ and the ratio of mobilities between the two channels $\gamma = \mu_2/\mu_1$. Errors peak when the conductivities of both channels are equal ($n_1\mu_1 = n_2\mu_2$).

band also exhibits a higher group velocity than the HH band, so the LH and HH carriers conduct in parallel, with different densities and mobilities. In such cases, $R_{xy}(B)$ and $R_{xx}(B)$ become nonlinear in B and must be analyzed with a Hall effect model that accounts for the plurality of carrier species.

This report applies a classical Drude model for isotropic magnetotransport of a two-carrier system. It then outlines a fitting procedure by which the total carrier density and mobility can be estimated from Hall effect measurements taken with sufficiently high magnetic field to capture nonlinear field dependence in $R_{xx}(B)$ and $R_{xy}(B)$. In the case where the two carriers populations have the same charge polarity ($q_1 = q_2$), we demonstrate that a single-carrier interpretation [Eqs. (1) and (2)] of low magnetic field measurements will always underestimate the total carrier concentration (thus overestimating the mobility). This is shown in Fig. 1 and discussed in Sec. II B.

This insight elucidates previous reports of temperature-dependent Hall effect measurements in (In)GaN/Al(GaN) 2DHGs,^{5,7,9,12–15} all of which utilize a single-carrier interpretation for low-field measurements, and report a roughly twofold decrease in hole density between room and cryogenic temperatures. In these reports, this temperature dependence is either ignored or temporarily attributed to dopant activation or temperature dependence of the GaN-air surface barrier height, piezoelectric constants, or spontaneous polarizations of the constituent materials. However, no model invoking these mechanisms has been adequate to account for this magnitude of change in carrier density. Using the two-carrier model, we demonstrate that this apparent reduction in carrier density can be explained by an increase in the LH-to-HH mobility ratio with decreasing temperature.

II. THE TWO-CARRIER TRANSPORT MODEL

The isotropic conductivity σ of m parallel-conducting carrier populations subject to a perpendicular magnetic field B can be expressed as¹⁶

$$\sigma_{xx}(B) = \sum_{i=1}^m \frac{\sigma_i}{1 + (\mu_i B)^2} = \sum_{i=1}^m \frac{q_i n_i \mu_i}{1 + (\mu_i B)^2}, \quad (3a)$$

$$\sigma_{xy}(B) = \sum_{i=1}^m \frac{\sigma_i(\mu_i B)}{1 + (\mu_i B)^2} = \sum_{i=1}^m \frac{q_i n_i \mu_i^2 B}{1 + (\mu_i B)^2}, \quad (3b)$$

with the convention that q_i and μ_i carry the same sign.¹⁷ From the conductivity tensor equation,

$$\mathbf{J} = \begin{pmatrix} \sigma_{xx} & \sigma_{xy} \\ -\sigma_{xy} & \sigma_{xx} \end{pmatrix} \mathbf{E} \iff \begin{pmatrix} R_{xx} & R_{xy} \\ -R_{xy} & R_{xx} \end{pmatrix} \mathbf{J} = \mathbf{E}, \quad (4)$$

the longitudinal and transverse resistances are expressed as

$$R_{xx}(B) = \frac{\sigma_{xx}(B)}{\sigma_{xx}^2(B) + \sigma_{xy}^2(B)}, \quad (5a)$$

$$R_{xy}(B) = \frac{\sigma_{xy}(B)}{\sigma_{xx}^2(B) + \sigma_{xy}^2(B)}. \quad (5b)$$

In the case of a single-carrier species ($m = 1$), Eqs. (1) and (2) are recovered. In the case of two carrier species, the resistance tensor components become

$$R_{xx} = \frac{\sigma_1(1 + (\mu_2 B)^2) + \sigma_2(1 + (\mu_1 B)^2)}{\sigma_1^2(1 + (\mu_2 B)^2) + 2\sigma_1\sigma_2(1 + (\mu_1\mu_2 B^2)) + \sigma_2^2(1 + (\mu_1 B)^2)}; \quad (6a)$$

$$R_{xy} = \frac{\sigma_1(\mu_1 B)(1 + (\mu_2 B)^2) + \sigma_2(\mu_2 B)(1 + (\mu_1 B)^2)}{\sigma_1^2(1 + (\mu_2 B)^2) + 2\sigma_1\sigma_2(1 + (\mu_1\mu_2 B^2)) + \sigma_2^2(1 + (\mu_1 B)^2)}. \quad (6b)$$

This tensor-style derivation of these equations is given in many places.^{16–19} Example use cases include parallel conduction in bulk and surface states,^{16,20} parallel conduction of electrons and holes,²¹ multi-sub-band occupation in quantum well heterostructures,²² and dual-band occupation in high-density n-type²³ and p-type²⁴ semiconductors. In any case, the mobility and density of each conducting channel are determined as free fitting parameters.

More complex Quantitative Mobility Spectrum Analysis (QMSA) algorithms have been developed^{25,26} which generalize Eq. (3) to account for non-uniform group velocity and carrier scattering within occupied bands. This is done by replacing the discrete densities n_i with “mobility spectra” $n(\mu)$ and making m large such that the summations approximate integration, significantly increasing the number of free fitting parameters. Such models have been applied previously to characterize magnetotransport in various n-type III-nitride semiconductor systems.^{27–29}

However, the simpler four-parameter two-carrier model employed in this paper proves adequate for distinguishing between HH and LH band transport. Further, this model is also shown to overfit measurements above ~ 100 K due to the lack of polynomial features in the fitted data; thus, employing a model with even more free parameters is unwarranted.

A. Polynomial expansion coefficients of $R_{xx}(B)$ and $R_{xy}(B)$

Valuable insight and restructuring of Eq. (6) is obtained by expanding in powers of B ,

$$R_{xx}(B) \approx \frac{\overbrace{1}^{c_0 = \frac{1}{\sigma} = R_{sh}}}{q_1 n_1 \mu_1 + q_2 n_2 \mu_2} + \frac{\overbrace{q_1 q_2 n_1 n_2 \mu_1 \mu_2 (\mu_1 - \mu_2)^2}^{c_2}}{(q_1 n_1 \mu_1 + q_2 n_2 \mu_2)^3} B^2 - \frac{\overbrace{q_1 q_2 n_1 n_2 \mu_1^3 \mu_2^3 (q_1 n_1 + q_2 n_2)^2 (\mu_1 - \mu_2)^2}^{c_4 = c_3^2 / c_2}}{(q_1 n_1 \mu_1 + q_2 n_2 \mu_2)^5} B^4 + \dots, \quad (7a)$$

$$R_{xy}(B) \approx \frac{\overbrace{q_1 n_1 \mu_1^2 + q_2 n_2 \mu_2^2}^{c_1 = R_H}}{q_1 n_1 \mu_1 + q_2 n_2 \mu_2} B - \frac{\overbrace{q_1 q_2 n_1 n_2 \mu_1^2 \mu_2^2 (\mu_1 - \mu_2)^2}^{c_3}}{(q_1 n_1 \mu_1 + q_2 n_2 \mu_2)^4} B^3 + \frac{\overbrace{q_1 q_2 n_1 n_2 \mu_1^4 \mu_2^4 (q_1 n_1 + q_2 n_2)^3 (\mu_1 - \mu_2)^2}^{c_5 = c_3^3 / c_2^2}}{(q_1 n_1 \mu_1 + q_2 n_2 \mu_2)^6} B^5 - \dots \quad (7b)$$

Here, $R_{xx}(B)$ and $R_{xy}(B)$ are seen to have strictly even and odd polynomial magnetic field dependence, respectively, with polynomial coefficients that alternate signs.

The c_j expansion coefficients in Eq. (7) satisfy a recursive condition,

$$c_{j \geq 2} = \frac{c_3^{j-2}}{c_2^{j-3}} = \frac{R_H^j}{R_{sh}^{j-1}} \frac{\beta(1-\beta)(\gamma-1)^2 \gamma^{j-1}}{\underbrace{(1+\beta(\gamma^2+1))^j}_{\sim 1}} \quad (8)$$

such that $R_{xx}(B)$ and $R_{xy}(B)$ in Eq. (6) are equivalently expressed (without truncating the expansions) as

$$R_{xx}(B) = c_0 + \frac{c_2 B^2}{1 + \left(\frac{c_3}{c_2} B\right)^2}, \quad (9a)$$

$$R_{xy}(B) = c_1 B - \frac{c_3 B^3}{1 + \left(\frac{c_3}{c_2} B\right)^2}. \quad (9b)$$

c_{0-3} are better-suited as fitting parameters for $R_{xx}(B)$ and $R_{xy}(B)$ measurements [than, for example, σ_1 , σ_2 , n_1 , and n_2 in Eq. (5)] for three reasons. (1) Each fitting parameter pertains to a unique polynomial feature of the data, minimizing covariance with c_0 or c_1 . (2) The covariance between c_2 and c_3 only influences how well the fit captures the $B^{>4}$ features of the $R_{xx}(B)$ and $R_{xy}(B)$ data [see Eq. (7)]. (3) This parameterization guards against over-fitting. If n_1 , n_2 , μ_1 , and μ_2 are not known *a priori*, then four free fitting parameters are required, and $R_{xx}(B)$ and $R_{xy}(B)$ must collectively exhibit four or more polynomial features to constrain these parameters uniquely. If the $c_3 B^3$ feature in $R_{xy}(B)$ is poorly resolved above the measurement noise, c_3 will be unconstrained, signaling over-fitting.

The band-resolved mobilities and densities are obtained from the c_{0-3} coefficients by

$$n_1 = \frac{q_1}{q_2} \frac{2c_2^3 - c_1 c_2 c_3 + c_3(c_0 c_3 + \text{sgn}(q_2) c^*)}{2q(c_1 c_3 - c_2^2) c^*}, \quad (10a)$$

$$n_2 = -\frac{2c_2^3 - c_1 c_2 c_3 + c_3(c_0 c_3 - \text{sgn}(q_2) c^*)}{2q(c_1 c_3 - c_2^2) c^*}, \quad (10b)$$

$$\mu_2 = \frac{c_1 c_2 + c_0 c_3 \mp \text{sgn}(q_2) c^*}{2c_0 c_2}, \quad (10c)$$

09 January 2025 13:58:58

where

$$c^* = \sqrt{c_2^2(c_1^2 + 4c_0c_2) - 2c_0c_1c_2c_3 + c_0^2c_3^2}. \quad (11)$$

These equations hold for any combination of signs in q_1 and q_2 .

In Sec. III A, this polynomial coefficient representation of the two-carrier model is applied to temperature-dependent magnetotransport measurements of a GaN/AlN 2DHG.

B. Error in single-carrier analysis of low-field Hall (for $q_1 = q_2$)

We now assess the nature of errors that arise from single-carrier Hall analysis of a system of two carriers with the same charge polarity ($q_1 = q_2$), as is the case for the GaN/AlN 2DHG discussed in Sec. III.

From the first-order term of $R_{xy}(B)$ in Eq. (7b) (with $q_1 = q_2$), we find that low-field Hall effect measurements (where B^{2+} terms are imperceptible) that are interpreted with a single-carrier model [see Eq. (2)] will infer an *apparent* carrier density of

$$n_{\text{apparent}} = \frac{1}{qR_H} = n \frac{(1 + \beta(\gamma - 1))^2}{1 + \beta(\gamma^2 - 1)}, \quad (12)$$

where $n = n_1 + n_2$ is the true carrier density, $\beta = n_2/n$ is the fraction of carriers in the 2nd channel, and $\gamma = \mu_2/\mu_1$ is the ratio between the two mobilities. The apparent mobility similarly differs according to

$$\mu_{\text{apparent}} = \frac{R_H}{R_{sh}} = \mu \frac{1 + \beta(\gamma^2 - 1)}{(1 + \beta(\gamma - 1))^2}, \quad (13)$$

where μ is the ensemble mobility,

$$\mu = \frac{\sigma}{qn} = \frac{\sigma_1 + \sigma_2}{q(n_1 + n_2)} = \frac{n_1\mu_1 + n_2\mu_2}{n_1 + n_2}. \quad (14)$$

The ratios n_{apparent}/n and $\mu_{\text{apparent}}/\mu$ are plotted in Fig. 1, as functions of β and γ . This plot demonstrates that single-carrier interpretations will always *under-estimate* total free carrier density and, consequently, *over-estimate* ensemble mobility. These errors magnify with increasing μ_2/μ_1 and are highest when $\sigma_1 = \sigma_2$. As expected, the measurement errors vanish for single-carrier-equivalent conditions: $\mu_2/\mu_1 = 1$, $n_2/n = 0$, or $n_2/n = 1$.

While this model is applied in this report to a degenerate hole gas (example band structure plotted in the inset of Fig. 1), the plotted errors apply to any system with isotropic parallel transport of two species with the same charge polarity.

III. MEASUREMENTS AND ANALYSIS

The sample investigated in this report consisted of a 10 nm thin film of undoped GaN on 700 nm of undoped AlN, grown by MBE on a single-crystal aluminum nitride substrate [see Fig. 2(c)]. The details of the early growth and transport studies can be found in Ref. 5, and the improved growth processes adopted in the samples in this study are discussed elsewhere.^{8,9,14} The GaN/AlN

interface is the only conducting layer in this entire material stack, as confirmed by previous studies³⁰ and illustrated in the energy band diagram in Fig. 2(c), calculated by self-consistent Poisson-k-p. Hence, the 2DHG is the only conducting channel (there is no buried 2DEG, such as in Ref. 12), and any signatures of parallel conduction in this sample must arise from the occupation of multiple hole sub-bands.

Magnetotransport measurements were performed in a Quantum Design PPMS[®] DynaCool[™] on a square as-grown sample with soldered indium contacts in a van der Pauw geometry. An excitation current of 100 μ A was used at temperatures from 2 to 390 K, sweeping magnetic field to ± 9 T. R_{xy} was measured along a single orientation, while R_{xx} was computed by the van der Pauw equation from two orthogonal measurements.¹⁹ $R_{xx}(B)$ measurements were symmetrized in B , and $R_{xy}(B)$ was anti-symmetrized to account for thermal gradients and contact misalignment.³¹ The measured magnetoresistance, $MR(B) = \frac{R_{xx}(B) - R_{xx}(0)}{R_{xx}(0)}$ and $R_{xy}(B)$, are plotted in Figs. 2(a) and 2(b), respectively.

A. Fitting $R_{xx}(B)$ and $R_{xy}(B)$ with the c_j coefficients

Measured $R_{xx}(B)$ and $R_{xy}(B)$ were fit to Eqs. (9a) and (9b), respectively, with c_{0-3} as free fitting parameters. Three different least-squares fits were performed (summarized in Table I):

- Measured $R_{xx}(B)$ were fit independently with c_0 , c_2 , and c_3 as free parameters (the “XX fit”).
- Measured $R_{xy}(B)$ were fit independently with c_1 , c_2 , and c_3 as free parameters (the “XY fit”).
- $R_{xx}(B)$ and $R_{xy}(B)$ were fit simultaneously with c_0 , c_1 , c_2 , and c_3 as free parameters (the “Simultaneous fit”).

The residuals between the model and the measured data points (B^i , R_{xx}^i , R_{xy}^i) were computed by

$$\chi_{xx}^i(c_0, c_2, c_3) = R_{xx}^i - R_{xx}(B_i; c_0, c_2, c_3), \quad (15a)$$

$$\chi_{xy}^i(c_1, c_2, c_3) = R_{xy}^i - R_{xy}(B_i; c_1, c_2, c_3). \quad (15b)$$

For each fit, the free parameters were varied to minimize the RMS residual χ^{RMS} (equations given in Table I). While all three fits should, in theory, yield equivalent band-resolved HH mobilities and densities, fitting $R_{xx}(B)$ and $R_{xy}(B)$ in isolation enables us to assess the degree to which the nonlinearity in each is described by parallel conduction vs other sources of magnetoresistance. Below 30 K, weak localization was observed, so measured resistances with $|B| < 0.5$ T were excluded from the fits.

The model results of the XX and XY fits are plotted against the measured data in Figs. 2(a) and 2(b), respectively. The corresponding values of χ_{xx}^{RMS} and χ_{xy}^{RMS} at each temperature are plotted in Fig. 2(c), ranging from ~ 10 to 100 Ω . Also plotted at each temperature is $c_j \times (9 \text{ T})^j$, the j th-order contribution to R_{xx} (even j) or R_{xy} (odd j) at $B = 9$ T. c_0 (or R_{sh}) increases from 0.33 $\text{k}\Omega/\square$ at 2 K to 5.1 $\text{k}\Omega/\square$ at room temperature. c_1 (or R_H) decreases slightly from $1/(2.3 \times 10^{13}) \text{ cm}^3/\text{q}$ at 2 K to $1/(4.4 \times 10^{13}) \text{ cm}^3/\text{q}$ at room temperature, consistent with the apparent decrease in carrier density seen in previous reports of GaN/AlN 2DHG

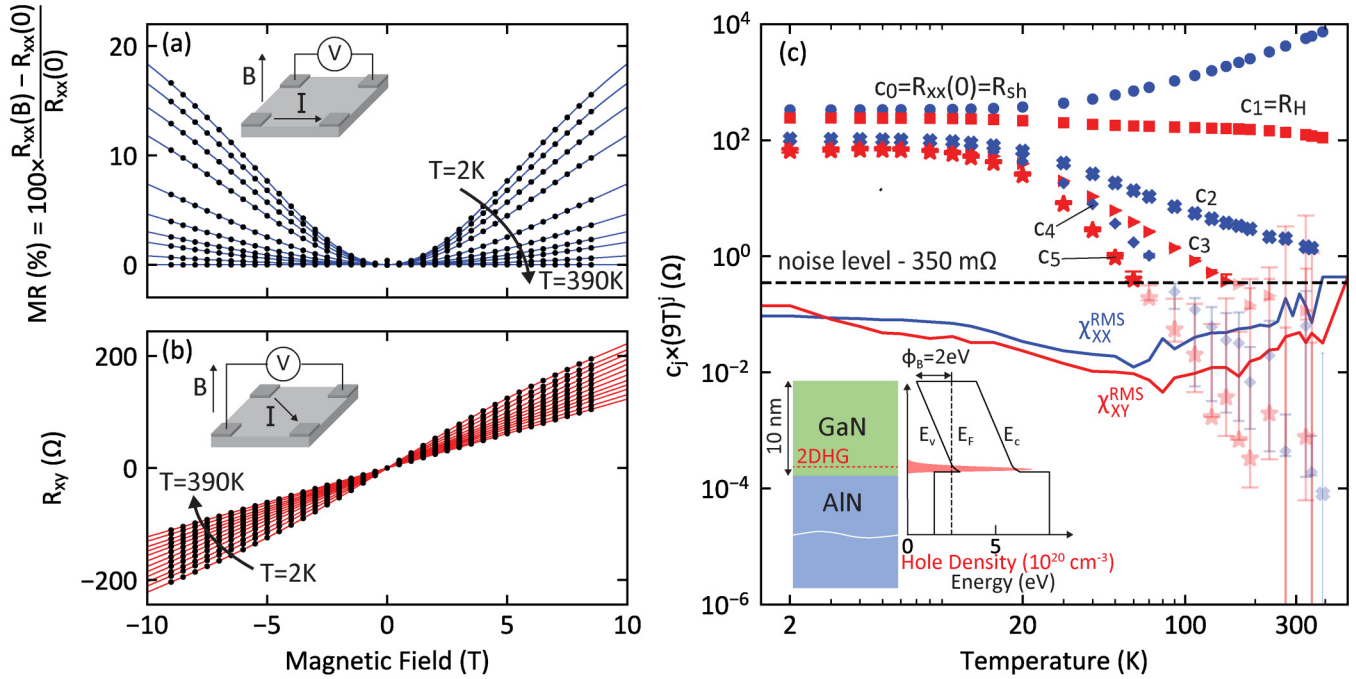


FIG. 2. Measured (a) $MR(B)$ and (b) $R_{xy}(B)$ of a GaN/AlN 2DHG, field-swept to $B = \pm 9$ T at select temperatures from 2 to 390 K. The four-point measurement configuration is shown for each. (c) The RMS residuals of the fits in (a) χ_{xx}^{RMS} (blue line) and (b) χ_{xy}^{RMS} (red line) vs temperature, and the signal contributions of the j th polynomial features of $R_{xx}(B)$ (even j) or $R_{xy}(B)$ (odd j) at $B = 9$ T (the maximum measured field) for $j = 0$ (circles), $j = 1$ (squares), $j = 2$ (Xs), $j = 3$ (triangles), $j = 4$ (diamonds), and $j = 5$ (stars). Points with an uncertainty bound below the 350mΩ noise line are dimmed. (Inset) GaN/AlN heterostructure, band diagram, and charge profile from self-consistent Poisson-k-p simulation.

transport.^{5,8,9,12} For all $j > 2$, the magnitude of the j th-order contribution is suppressed monotonically with increasing temperature in conjunction with the increase in R_{sh} [see Eq. (8)].

Below 350 mΩ, c_3 ceases changing monotonically with temperature, as anticipated from Eq. (8) and the observed trend in c_0 . Fits at each temperature are only considered valid if the lower uncertainty bound of all four free parameters lies above this noise line, as any nonlinearity observed in $R_{xy}(B)$ is not otherwise convincingly attributable to two-carrier phenomena. The three fits are, thus, cut off between 70 and 170 K.

The error bars in Fig. 2(c) indicate the range of parameters for which χ^{RMS} differs from the minimum by $\leq 10\%$. This uncertainty range is for larger j and increases at high temperatures where these polynomial features lose prominence in the measured data. The error bounds are discussed further in Sec. III C.

B. The HH and LH densities and mobilities vs temperature

Figure 3 shows the GaN HH and LH ($q_1 = q_2 = +q$) (a) densities ($p_{HH} = n_1, p_{LH} = n_2$) and (b) mobilities ($\mu_{HH} = \mu_1, \mu_{LH} = \mu_2$),

TABLE I. Specifications for the three least-squares fits performed in this study.

Fit name	Color	Fitted data	Free parameters	Constrained parameters	Minimized quantity
XX	●	(B^i, R_{xx}^i)	c_0, c_2, c_3	c_1 —from XY	$\chi_{xx}^{RMS} = \sqrt{\frac{\sum_i \chi_{xx}^i(c_0, c_2, c_3)^2}{N-3}}$
XY	●	(B^i, R_{xy}^i)	c_1, c_2, c_3	c_0 —from XX	$\chi_{xy}^{RMS} = \sqrt{\frac{\sum_i \chi_{xy}^i(c_1, c_2, c_3)^2}{N-3}}$
Simultaneous	●	$(B^i, R_{xx}^i, R_{xy}^i)$	c_0, c_1, c_2, c_3	...	$\chi_{sim}^{RMS} = \sqrt{\frac{\sum_i \chi_{xx}^i(c_0, c_2, c_3)^2 + \chi_{xy}^i(c_1, c_2, c_3)^2}{2N-4}}$

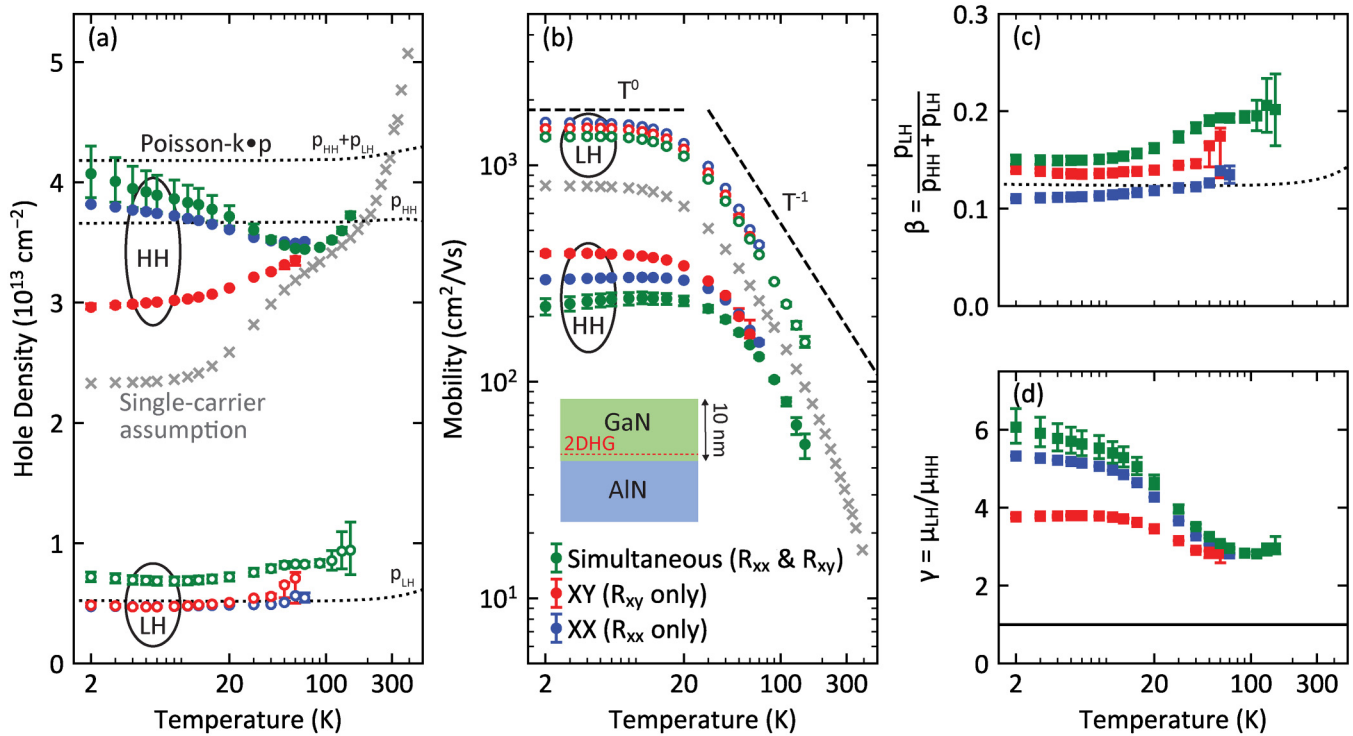
and the corresponding values of (c) $\beta = p_{LH}/(p_{HH} + p_{LH})$ and (d) $\gamma = \mu_{LH}/\mu_{HH}$ obtained from the XX fit (blue), XY fit (red), and simultaneous fit (green). As in Fig. 2(c), error bars indicate the bounds of the parameter space in which χ^{RMS} is within 10% of the minimum. Each fit is plotted until the temperature at which the lower bound of one of the free parameters lies below the noise line (350 mΩ).

In Fig. 3(a), the band-resolved densities are corroborated against a self-consistent Poisson and multi-band k-p simulation of the heterostructure's band profile and valence band structure. The details of this model are discussed elsewhere.^{11,32} Input parameters were obtained from Ref. 33, accounting for temperature dependence in the lattice³⁴ and elastic³⁵ constants of GaN and AlN. The fermi level on the GaN surface is set to 2 eV above the valence band, in accordance with contactless electro-reflectance measurements of similar GaN/AlN 2DHG samples.³⁰ The simulated band-resolved carrier densities are plotted in Figs. 3(a) and 3(c).

The XX and XY fits both report $p_{LH} \sim 0.5 \times 10^{13} \text{ cm}^{-2}$, in agreement with the simulation. p_{HH} from the XY fit increases from $3.0 \times 10^{13} \text{ cm}^{-2}$ at 2 K to $3.4 \times 10^{13} \text{ cm}^{-2}$ at 90 K, while p_{HH} from the XX and simultaneous fits decreases to the same from $\sim 4 \times 10^{13} \text{ cm}^{-2}$ at 2 K. The discrepancy in the results of these fitting schemes points to additional sources of magnetotransport not captured in this two-carrier model.

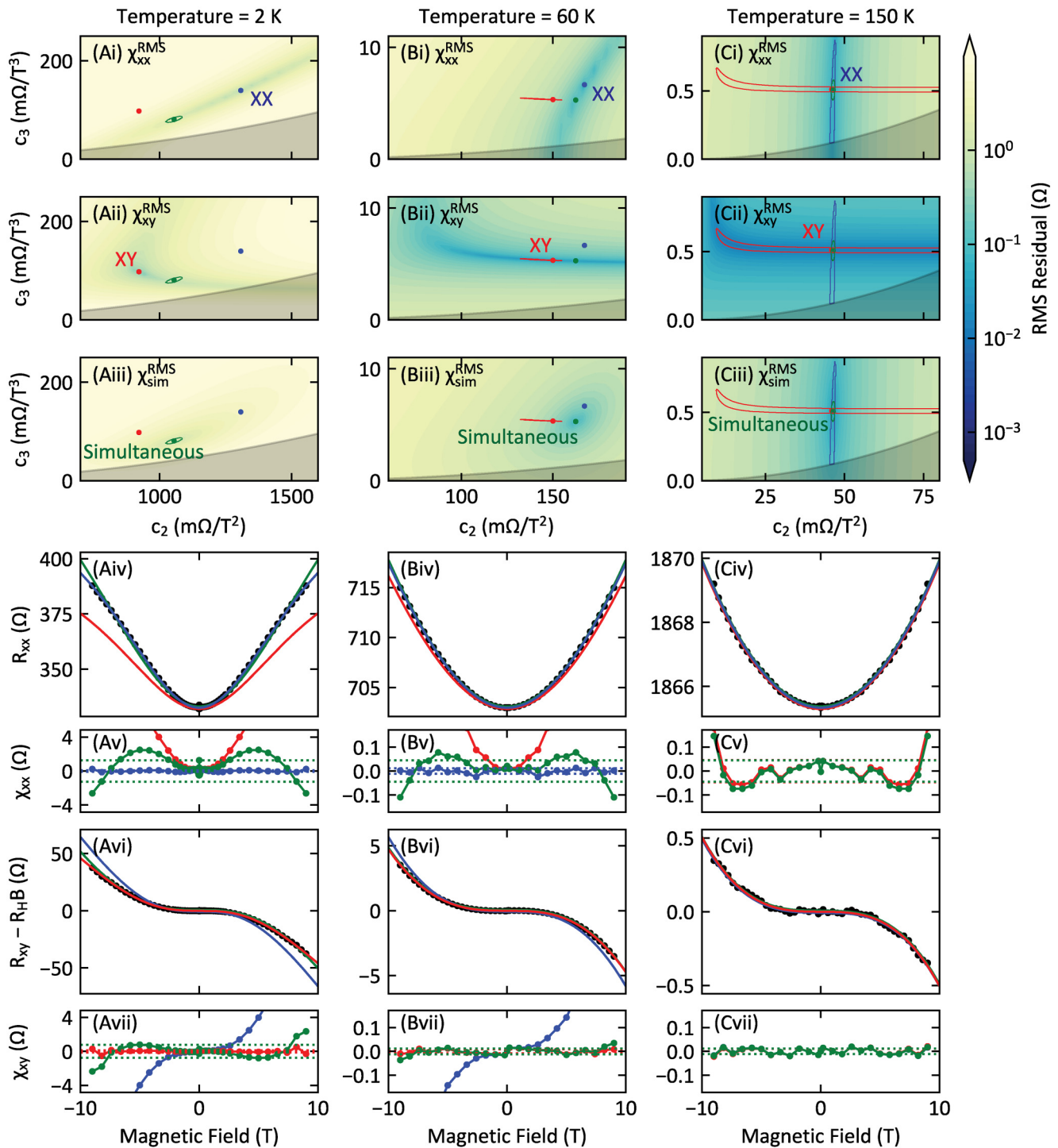
By contrast, the carrier density inferred from single-carrier analysis of the low-field resistances (open gray circles) decreases by almost 50% between 300 and 2 K, similar to the Hall results reported previously for various GaN 2DHGs.^{5,7,9,12-15} This drop coincides with a roughly twofold increase in γ over the same temperature range. Inspecting Fig. 1 and Eq. (7b), such an increase in γ at fixed β would produce an anomalous decrease in the apparent carrier density. Thus, we assert that the previously reported temperature-dependent carrier density below room temperature is fully explainable as an artifact of parallel conduction.

Above room temperature, however, the density inferred from single-carrier analysis increases to $\sim 5 \times 10^{13} \text{ cm}^{-2}$ near 400 K, well above the simulated value. Reference 15 reports a similar approximately linear increase in the carrier density of a GaN/AlN 2DHG with increasing temperature, reaching $\sim 1 \times 10^{14} \text{ cm}^{-2}$ near 800 K, albeit using a single-carrier interpretation. Under the assumptions of the model derived in this report, the single-carrier density is *always* an underestimate of the total density [per Eq. (12) and Fig. 1]. Thus, these measurements imply an *actual* increase in the GaN 2DHG density above room temperature. As mentioned earlier, no definitive explanation yet exists as to a temperature-dependent source or sink of charges. However, understanding the origin of these carriers may prove crucial for the future utilization of GaN 2DHGs in high-temperature applications.



09 January 2025 13:58:58

FIG. 3. Measured LH (open circles) and HH (closed circles) (a) carrier density and (b) mobility, (c) $\beta = p_{LH}/(p_{HH} + p_{LH})$, and (d) $\gamma = \mu_{LH}/\mu_{HH}$ ratios from two-carrier analysis by least-squares fitting of $R_{xx}(B)$ (blue) and $R_{xy}(B)$ (red) measurements, and both simultaneously (green). Gray Xs denote the total density and mobility inferred from a single-carrier fitting of the low-field measurements. Black dotted lines denote Poisson-k·p simulation of the band-resolved densities. T^0 and T^{-1} power laws are plotted in (b) for reference.



09 January 2025 13:58:58

FIG. 4. Demonstration of least-squares fitting at representative temperatures: (a) 2, (b) 60, and (c) 150 K. Density plots in (c_2, c_3) space of the RMS residual of the two-carrier model against (i) $R_{xx}(B)$ and (ii) $R_{xy}(B)$ measurements, and (iii) both simultaneously. Contours enclose the regions in which χ^{RMS} is within 10% of the minimum when fitting $R_{xx}(B)$ (blue), $R_{xy}(B)$ (red), and both simultaneously (green). (iv) Measured $R_{xx}(B)$ and (vi) $R_{xy}(B) - R_H B$ and the associated model results for the best-fit parameters in (i)–(iii), with the corresponding points-wise (connected points) and RMS (horizontal dotted line) residuals plotted in (v) and (vii), respectively.

The LH and HH mobilities in Fig. 3(b) saturate at temperatures below 20 K, indicating extrinsically limited transport. Between the three fits, HH mobilities measured at 2 K range from 222 to 393 cm²/Vs, and LH mobilities range from 1350 to 1580 cm²/Vs. In Fig. 3(d), the μ_{LH}/μ_{HH} ratio saturates between 3.7 and 6.1 near 10 K.

Above 20 K, the mobilities are phonon-limited and decrease with a $\sim T^{-1}$ power law. This trend is consistent with simulations of hole mobility in GaN/AlN 2DHGs¹¹ which show that room temperature transport is primarily limited by acoustic deformation potential (ADP) scattering, which goes as $\mu_{2d}^{ADP} \sim ((m^*)^2 k_B T)^{-1}$ in two dimensions^{36,37} (m^* is the effective mass). In the same temperature range, the mobility ratio [in Fig. 3(d)] approaches $\gamma = \mu_{LH}/\mu_{HH} \sim 2.9$. Although γ is not discernable at room temperature due to the suppression of $j > 2$ polynomial features, this saturation suggests that γ will be greater than 1. Thus, even if $R_{xy}(B)$ appears linear within the measured magnetic field range, single-carrier estimates of the ensemble hole density and mobility are still subject to the errors plotted in Fig. 1.

It is worth noting here that we have explicitly assumed a unity Hall factor $r_H = \mu_H/\mu_d$, which relates the Hall (μ_H) and drift (μ_d) mobilities. In the case of a single parabolic band ($E_k = \hbar^2 k^2/2m^*$) and scattering mechanisms of the form $\tau_k = \tau_0 (\frac{E_k}{k_B T})^p$ (τ_0 is a characteristic scattering time; k_B is Boltzmann's constant), the Hall factor in a two-dimensional system is given by^{37,38}

$$r_H = \frac{\Gamma(2p+2) F_{2p}(\eta) F_0(\eta)}{\Gamma^2(p+2) F_p^2(\eta)}, \quad (16)$$

where $\eta = E_F/k_B T$ and $F_j(\eta) = \frac{1}{\Gamma(j+1)} \int_0^\infty \frac{u^j du}{1+e^{u-\eta}}$ is the Fermi-Dirac integral. r_H reduces to unity both in the $\eta \gg 1$ limit, valid below ~ 50 K for this system, and when $p = 0$, which is the form of ADP scattering in a two-dimensional system,^{36,37} which is the dominant scattering mechanism in this system above ~ 50 K.¹¹ However, the temperature-dependent Hall factors for each band in this system, accounting for all scattering mechanisms, interband scattering, band-nonparabolicity, etc., have not been modeled to date.

C. Fitting discrepancies and error analysis

The nature of the discrepancy between the XX, XY, and simultaneous fit results and the size of the error bounds on the fit parameters are demonstrated in Fig. 4. This discrepancy manifests in the differing determinations of n_1 , n_2 , μ_1 , and μ_2 seen at low temperature in Fig. 3 and points to physics not contained in the assumptions of the model.

Model fittings at representative temperatures $T = 2, 60,$ and 150 K are plotted in columns A–C, respectively. The top rows of the figure show heat-density plots of χ^{RMS} (equations given in Table I) for the (i) XX fit, (ii) XY fit, and (iii) simultaneous fit. c_0 and c_1 are equal in all plots at a given temperature, while c_2 and c_3 are varied to encompass the three best-fit regions. The plotted contours encompass all points in (c_2, c_3) , where χ^{RMS} is within 10% of the minimum. The darkened area in the lower-right corner of these plots denotes the region below $c_3 = c_2^2/c_1$, where the band-resolved densities change sign [see the denominator in Eqs. (10a) and (10b)].

Row (iv) plots the measured $R_{xx}(B)$ and associated model result [Eq. (9a)] for the best-fit parameters in the plots above. The point-wise (connected points) and RMS (dotted line) residuals for each fit are plotted vs magnetic field in row (v). Row (vi) similarly plots the measured and modeled $R_{xy}(B) - R_H B$, with the residuals plotted below in (vii).

The region of best-fit (lowest χ_{xx}^{RMS}) for R_{xx} (row i) appears as a diagonal line in the (c_2, c_3) space at 2 K but becomes vertical at higher temperature as the 4th-order features are suppressed and cannot dictate the value of c_3 [by Eq. (8)]. Consequently, the contour of best fit in Fig. 3(Ci) becomes large, as do the error bars on c_4 in Fig. 2(c).

Similarly, the best-fit (lowest χ_{xy}^{RMS}) region for R_{xy} (row ii) draws an “L”-shaped contour in the (c_2, c_3) space. At lower temperatures, the XY best-fit sits higher on the “L,” indicating a strong $c_3 B^5$ signal [again, see Eq. (8)], while at higher temperatures, the XY best-fit sits in the horizontal region.

At 2 K, the XX and XY best-fit regions differ by $\sim 20\%$ in c_2 and c_3 . This discrepancy is seen visually in the model fits (Aiv, Avii) and the trend in the residuals of the XY fit against the R_{xx} data (in Av), and the residuals of the XX fit against the R_{xy} data (in Avii). Consequently, the simultaneous fit is an imperfect representation of both the $R_{xx}(B)$ and $R_{xy}(B)$ measurements, and a trend is seen in the residuals of both data sets (see Av and Avii).

In the supplementary material, Figs. 2–4 are recreated for a different GaN/AlN 2DHG sample capped with Mg-doped p-type GaN and measured in a Hall bar geometry. The same qualitative disagreement between the XX, XY, and simultaneous fits is observed in these measurements, which signals that the discrepancy between the XX and XY fits arises from additional physics not accounted for in the present model, such as inter-band scattering between the LH and HH bands²² or weak localization.

IV. CONCLUSIONS

The large polarization-induced hole density at the GaN/AlN hetero-interface degenerately occupies both the HH and LH bands of the GaN; thus, Hall effect measurements must be interpreted using a proper multi-carrier model.

This work describes an isotropic Drude model of two parallel-conducting carrier populations with the same charge polarity. We apply this model to fit 2DHG transport measurements from 2 to 170 K across a magnetic field range of ± 9 T. We successfully resolve HH and LH carrier densities that are of comparable magnitude to self-consistent Poisson-k-p simulations, as well as band-resolved mobilities, observing cryogenic mobilities of $\mu_{HH} \sim 300$ and $\mu_{LH} \sim 1400$ cm²/Vs. Although it does not represent ensemble conduction, this LH mobility is the largest hole mobility reported to date in GaN.

The three fitting approaches presented in this study (XX, XY, and simultaneous) provide a helpful gauge for the adequacy of the two-carrier model to explain the magnetotransport in the system under test. The level of agreement between the mobilities and densities extracted from the XX and XY fits indicates the degree to which parallel conduction adequately explains the magnetotransport. In the case where they disagree (e.g., below 20 K in this study), a more complicated physical model with additional

parameters is needed. The simultaneous fit allows one to extend the two-carrier fitting into regimes (e.g., up to 170 K in this study) where $R_{xx}(B)$ and $R_{xy}(B)$ have insufficient polynomial features on their own to constrain the XX or XY fits, but collectively have four polynomial features. We suggest that all three fitting approaches be exercised for a given problem to evaluate the accuracy and error bounds of the two-carrier modeling procedure outlined in this study.

These results offer a credible experimental signature of LH band occupation in GaN and elucidate previously reported temperature-dependent transport results in GaN 2DHGs, which suggested an unexpected decrease in carrier density with decreasing temperature. The polarization-doped high-density p-type GaN employed in this study, which has only become available in recent years,⁵ provides an unprecedented platform to interrogate the valence band properties of GaN.

In future transport studies of GaN 2DHGs (or similar systems with multiple parallel-conducting channels), if the sample's sheet resistance is too large or sufficiently large magnetic fields are not available to observe higher-order polynomial features in $R_{xx}(B)$ and $R_{xy}(B)$, transport simulations should predict and be corroborated against measurements of R_{sh} and R_H , which are directly measurable, rather than n and μ .

SUPPLEMENTARY MATERIAL

In the [supplementary material](#), Figs. 2–4 are recreated for magnetotransport measurements of a GaN/AlN 2DHG sample with a Mg-doped p-type GaN cap, measured in a Hall bar geometry up to ± 9 T from 3 to 390 K.

ACKNOWLEDGMENTS

This work was supported in part by SUPREME (modeling), one of seven centers in JUMP 2.0, a Semiconductor Research Corporation (SRC) program sponsored by DARPA, and as part of the Ultra Materials for a Resilient Energy Grid (epitaxy and device fabrication), an Energy Frontier Research Center funded by the US Department of Energy, Office of Science, Basic Energy Sciences under Award No. DESC0021230 and in part by the ARO Grant No. W911NF2220177 (characterization). The authors acknowledge the use of facilities and instrumentation supported by NSF through the Cornell University Materials Research Science and Engineering Center DMR-1719875, and the use of the Cornell NanoScale Facility (CNF), a member of the National Nanotechnology Coordinated Infrastructure (NNCI), which was supported by the National Science Foundation (NSF Grant No. NNCI-2025233). The authors benefited from discussions with Jie-Cheng Chen, Feliciano Giustino, Changkai Yu, Samuel James Bader, and Reet Chaudhuri.

AUTHOR DECLARATIONS

Conflict of Interest

The authors have no conflicts to disclose.

Author Contributions

J. E. Dill and C. F. C. Chang, contributed equally to this work.

Joseph E. Dill: Investigation (equal); Methodology (equal); Visualization (equal); Writing – original draft (equal). **Chuan F. C. Chang:** Conceptualization (equal); Data curation (equal); Investigation (equal); Writing – review & editing (equal). **Debdeep Jena:** Funding acquisition (equal); Supervision (supporting); Writing – review & editing (equal). **Huili Grace Xing:** Funding acquisition (equal); Supervision (equal); Writing – review & editing (equal).

DATA AVAILABILITY

The data that support the findings of this study are available from the corresponding authors upon reasonable request.

REFERENCES

- 1M. S. Shur, A. D. Bykhovskii, R. Gaska, J. W. Yang, G. Simin, and M. A. Khan, *Appl. Phys. Lett.* **76**, 3061 (2000).
- 2W. Kim, A. Salvador, A. E. Botchkarev, O. Aktas, S. N. Mohammad, and H. Morcoç, *Appl. Phys. Lett.* **69**, 559 (1996).
- 3W. Götz, N. M. Johnson, J. Walker, D. P. Bour, and R. A. Street, *Appl. Phys. Lett.* **68**, 667 (1996).
- 4D. J. Kim, D. Y. Ryu, N. A. Bojarczuk, J. Karasinski, S. Guha, S. H. Lee, and J. H. Lee, *J. Appl. Phys.* **88**, 2564 (2000).
- 5R. Chaudhuri, S. J. Bader, Z. Chen, D. A. Muller, H. G. Xing, and D. Jena, *Science* **365**, 1454 (2019).
- 6O. Ambacher, J. Smart, J. R. Shealy, N. G. Weimann, K. Chu, M. Murphy, W. J. Schaff, L. F. Eastman, R. Dimitrov, L. Wittmer, M. Stutzmann, W. Rieger, and J. Hilsenbeck, *J. Appl. Phys.* **85**, 3222 (1999).
- 7C. Beckmann, J. Wieben, T. Zweipfennig, A. Kirchbrücher, J. Ehrler, R. Stamm, Z. Yang, H. Kalisch, and A. Vescan, *J. Phys. D: Appl. Phys.* **55**, 435102 (2022).
- 8R. Chaudhuri, Z. Chen, D. A. Muller, H. G. Xing, and D. Jena, *J. Appl. Phys.* **130**, 025703 (2021).
- 9Z. Zhang, J. Encomendero, R. Chaudhuri, Y. Cho, V. Protasenko, K. Nomoto, K. Lee, M. Toita, H. G. Xing, and D. Jena, *Appl. Phys. Lett.* **119**, 162104 (2021).
- 10E. H. Hall, *Am. J. Math.* **2**, 287 (1879).
- 11S. J. Bader, R. Chaudhuri, M. F. Schubert, H. W. Then, H. G. Xing, and D. Jena, *Appl. Phys. Lett.* **114**, 253501 (2019).
- 12P. Liu, K. Kakushima, H. Iwai, A. Nakajima, T. Makino, M. Ogura, S. Nishizawa, and H. Ohashi, "Characterization of two-dimensional hole gas at GaN/AlGaIn heterointerface," in *The 1st IEEE Workshop on Wide Bandgap Power Devices and Applications* (IEEE, 2013), pp. 155–158.
- 13G. Li, R. Wang, B. Song, J. Verma, Y. Cao, S. Ganguly, A. Verma, J. Guo, H. G. Xing, and D. Jena, *IEEE Electron Device Lett.* **34**, 852 (2013).
- 14R. Chaudhuri, Z. Zhang, H. G. Xing, and D. Jena, *Adv. Electron. Mater.* **8**, 2101120 (2022).
- 15R. Chaudhuri, *Integrated Electronics on Aluminum Nitride* (Springer, 2022), p. XVI, 255.
- 16R. L. Petritz, *Phys. Rev.* **110**, 1254 (1958).
- 17J. S. Kim, D. G. Seiler, and W. F. Tseng, *J. Appl. Phys.* **73**, 8324 (1993).
- 18N. W. Ashcroft and N. D. Mermin, *Solid State Physics* (Holt-Saunders, 1976), p. 240.
- 19F. Werner, *J. Appl. Phys.* **122**, 135306 (2017).
- 20C. Czaya, K. Keck, and H. Rühllein, *Phys. Status Solidi B* **82**, 595 (1977).
- 21D. E. Soule, *Phys. Rev.* **112**, 698 (1958).
- 22E. Zaremba, *Phys. Rev. B* **45**, 14143 (1992).
- 23M. A. Khan, J. N. Kuznia, J. M. Van Hove, N. Pan, and J. Carter, *Appl. Phys. Lett.* **60**, 3027 (1992).
- 24B. W. Kim and A. Majerfeld, *J. Appl. Phys.* **79**, 1939 (1996).
- 25W. A. Beck and J. R. Anderson, *J. Appl. Phys.* **62**, 541 (1987).
- 26W. A. Beck, *J. Appl. Phys.* **129**, 165109 (2021).

- ²⁷N. Biyikli, J. Xie, Y.-T. Moon, F. Yun, C.-G. Stefanita, S. Bandyopadhyay, H. Morkoç, I. Vurgaftman, and J. R. Meyer, *Appl. Phys. Lett.* **88**, 142106 (2006).
- ²⁸T. B. Fehlberg, G. A. Umana-Membreno, B. D. Nener, G. Parish, C. S. Gallinat, G. Koblmuller, S. Bernardis, and J. S. Speck, in *2006 Conference on Optoelectronic and Microelectronic Materials and Devices* (IEEE, 2006), pp. 11–14.
- ²⁹S. B. Lisesivdin, S. Acar, M. Kasap, S. Ozcelik, S. Gokden, and E. Ozbay, *Semicond. Sci. Technol.* **22**, 543 (2007).
- ³⁰L. Janicki, R. Chaudhuri, S. J. Bader, H. G. Xing, D. Jena, and R. Kudrawiec, *Phys. Status Solidi RRL* **15**, 2000573 (2021).
- ³¹D. C. Look, *Electrical Characterization of GaAs Materials and Devices* (Wiley, New York, 1989).
- ³²S. Birner, “Modeling of semiconductor nanostructures and semiconductor–Electrolyte interfaces,” Ph.D. thesis (Technische Universität München, 2011).
- ³³I. Vurgaftman and J. R. Meyer, *J. Appl. Phys.* **94**, 3675 (2003).
- ³⁴S. Figge, H. Kröncke, D. Hommel, and B. M. Epelbaum, *Appl. Phys. Lett.* **94**, 1 (2009).
- ³⁵Y. Dai, S. Li, H. Gao, W. Wang, Q. Sun, Q. Peng, C. Gui, Z. Qian, and S. Liu, *J. Mater. Sci. Mater. Electron.* **27**, 2004 (2016).
- ³⁶J. H. Davies, *The Physics of Low-dimensional Semiconductors* (Cambridge University Press, 1997).
- ³⁷D. Jena, *Quantum Physics of Semiconductor Materials and Devices* (Oxford University Press, Oxford, 2022), p. 655.
- ³⁸Y. Hasegawa, Y. Ishikawa, T. Saso, H. Shirai, H. Morita, T. Komine, and H. Nakamura, *Physica B* **382**, 140 (2006).